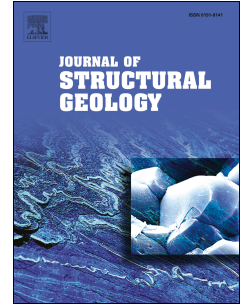


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L. Nègre, H. Stünitz, H. Raimbourg, A. Lee, J. Précigout, P. Pongrac, P. Jeřábek



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## **Author statement**

All authors have read the manuscript and agree with its content.

Journal Pre-proof

# 1 **Effect of pressure on the deformation of quartz aggregates in** 2 **the presence of H<sub>2</sub>O**

3 Authors: L. Nègre <sup>a,\*</sup>, H. Stünitz <sup>a,b</sup>, H. Raimbourg <sup>a</sup>, A. Lee <sup>b</sup>, J. Précigout <sup>a</sup>, P.  
4 Pongrac <sup>c</sup>, P. Jeřábek <sup>c</sup>

5 <sup>a</sup> Institut des Sciences de la Terre d'Orléans (ISTO), UMR 7327, CNRS/BRGM, Université  
6 d'Orléans, 45071 Orléans, France

7 <sup>b</sup> Department of Geology, University of Tromsø, Dramsveien 201, 9037 Tromsø, Norway

8 <sup>c</sup> Institute of Petrology and Structural Geology, Faculty of Science, Charles University in Prague,  
9 Albertov 6, 128 43 Prague, Czech Republic

10

11 Email addresses: lucille.negre@univ-orleans.fr (L. Nègre), holger.stunitz@uit.no (H.  
12 Stünitz), hugues.raimbourg@univ-orleans.fr (H. Raimbourg), amicia.lee@uit.no (A.  
13 Lee), jacques.precigout@univ-orleans.fr (J. Précigout), petar.pongrac@natur.cuni.cz  
14 (P. Pongrac), jerabek.petr@natur.cuni.cz (P. Jeřábek)

15

16 Corresponding author: Institut des Sciences de la Terre d'Orléans (ISTO), 1a rue  
17 de la Férollerie, 45100 Orléans, France, +332 38 25 50 27

18

19 Keywords: quartz deformation, quartz rheology, H<sub>2</sub>O weakening, dynamic  
20 recrystallization

21 **Abstract**

22 Quartzite samples of high purity with a grain size of  $\sim 200 \mu\text{m}$  have been  
23 experimentally deformed by coaxial shortening in a solid medium apparatus at  $900^\circ\text{C}$   
24 and at confining pressures ranging from 0.6 to 2 GPa. Most samples have been  
25 shortened by  $\sim 30\%$  with 0.1 wt.% added  $\text{H}_2\text{O}$ . The samples deformed dominantly by  
26 crystal plasticity (dislocation creep), and there is a systematic decrease of flow stress  
27 with increasing confining pressure. Strain rate stepping tests yield stress exponents  
28 of  $n \sim 1.4$ . The strain determined from individual grain shapes matches that  
29 determined from bulk shortening. In addition to plastic strain, mode I cracks  
30 developed in all samples, principally in the grain boundary regions. Recrystallized  
31 material, visible through cathodoluminescence colours, forms by two mechanisms:  
32 (1) progressive subgrain rotation and (2) cracking, nucleating small new grains. After  
33 high-angle boundaries have been established, grain boundary migration takes place,  
34 and a distinction of new grains nucleation origin (subgrain rotation or cracking) is  
35 impossible. At higher pressure, there is more recrystallized material forming in the  
36 deformed samples, and it is inferred that the inverse pressure dependence of flow  
37 stress is caused by enhanced grain boundary migration at higher pressure,  
38 consistent with previous studies.

## 39 1. Introduction

40 Despite a long history of research since the early discovery of the effect of H<sub>2</sub>O on  
41 quartz rheology by Griggs and Blacic (1965), many aspects of H<sub>2</sub>O weakening in  
42 quartz are yet unresolved. One of these aspects is the clearly observed dependence  
43 of the H<sub>2</sub>O weakening on pressure: it had been suggested by Paterson and  
44 Kekulawala, (1979), Tullis et al. (1979), Blacic (1981) and Mainprice and Paterson  
45 (1984) that the confining pressure has an enhancing effect on the H<sub>2</sub>O weakening  
46 effect in quartz, speculating that pressure may positively affect the diffusive uptake of  
47 H<sub>2</sub>O. The seminal work by Kronenberg and Tullis (1984) clearly demonstrated and  
48 quantified the increasing weakening of quartzite with increasing confining pressure in  
49 the presence of H<sub>2</sub>O. However, not much later it has been shown by Kronenberg et  
50 al. (1986) and Gerretsen et al. (1989) that the diffusion of H<sub>2</sub>O into quartz occurs at  
51 rates far too slow (or not at all) to play a role in deformation experiments, and that  
52 the infiltration of H<sub>2</sub>O into quartz is achieved by microcracking in experiments. Thus,  
53 the effect of pressure on deformation of quartz in the presence of H<sub>2</sub>O had been  
54 demonstrated but its cause was not very clear. Tullis and Yund (1989) have  
55 presented a study of annealing deformed quartz crystals at different pressures in the  
56 presence of H<sub>2</sub>O and have shown that recovery and recrystallization are enhanced  
57 by higher pressures.

58 In more recent studies of fine grained quartz material Rutter and Brodie (2004a,b)  
59 have determined stress exponents of  $n \approx 3$  and  $n = 1$  and have demonstrated grain  
60 size sensitive flow in quartz. These  $n$ -values are lower than previous  $n$ -values of  $n =$   
61 4 determined by Paterson and Luan (1990) and Luan and Paterson (1992) for  
62 dislocation creep. Fukuda et al. (2018) and Richter et al. (2018) have determined  $n$ -  
63 values of 1.7 to 1.9, respectively. These values were partly interpreted as a  
64 combination of dislocation and diffusion creep mechanisms. As the experiments of

65 Kronenberg and Tullis (1984) have been carried out using fine grained novaculite,  
66 and as this material has also been used in the experiments by Fukuda et al. (2018),  
67 it appears likely that the pressure dependence of deformation has been determined  
68 in material partially deformed by diffusion creep processes in the presence of H<sub>2</sub>O.  
69 The solubility of quartz in supercritical H<sub>2</sub>O increases non-linearly with increasing  
70 pressure (Manning, 1994, 2018), so that solution-precipitation processes may be  
71 expected to be enhanced at high pressures. Diffusion creep in the presence of H<sub>2</sub>O  
72 will involve dissolution and precipitation and therefore is expected to be pressure  
73 dependent. The pressure dependence of quartz rheology could therefore arise from  
74 enhanced dissolution at higher pressures. For this reason, we have revisited the  
75 pressure dependence of quartz deformation using a coarser grained quartzite (~200  
76 μm grain size) as starting material in order to diminish the role of diffusion creep  
77 processes and to test whether the dependence of quartz strength on pressure is the  
78 same as in fine grained material and to study the processes of weakening in more  
79 detail.

## 80 **2. Methods**

### 81 *2.1 Deformation experiments*

82 Coaxial shortening deformation experiments of natural quartzite (for the  
83 characterization of the starting material see the results-section) have been performed  
84 in two types of solid medium apparatus (a modified Griggs apparatus at University of  
85 Tromsø, Norway and a new generation solid medium apparatus at ISTO, Orléans,  
86 France). NaCl was used as the confining medium in both sets of experiments. The  
87 preparation procedure followed for samples deformed in Orléans corresponds to the  
88 one described by Précigout et al. (2018). For samples deformed in Tromsø,  
89 assemblies differ in size (smaller diameter and length) and use a single  
90 thermocouple instead of two.

91 Cylindrical samples (diameter 6.35 mm for samples in the conventional Griggs  
92 apparatus in Tromsø and 8 mm for the new rig in Orléans) were cored from the  
93 starting material (without any preferential orientation), with ends ground flat and  
94 plane-parallel to  $\sim 0.001$  mm tolerance. They were dried at  $110^{\circ}\text{C}$  for one day  
95 (minimum), then weighed, wrapped with two layers of nickel foil (0.025 mm  
96 thickness), inserted into platinum jacket (0.15 mm thickness) and weld-sealed after  
97 addition of 0.1 wt.% of distilled water.

98 For deformation experiments at 2 GPa confining pressure, the sample assembly  
99 was slightly modified. The top piston and top inner salt piece were separated into two  
100 parts and a disc of compressed salt (of approximately 2.6 mm thickness and 13 mm  
101 diameter) was inserted in between. In this way, the deformation column was able to  
102 compact as P and T were increased to the final P-T conditions of the experiment  
103 without applying much deviatoric stress on the sample when advancing the pistons.  
104 Before this modification of the sample assembly, several experiments failed as the  
105 hit-point was reached during the pumping stage and the sample started to deform  
106 earlier than expected.

107 All deformation experiments have been performed to approximately 30% coaxial  
108 shortening (total strain of samples is between  $\sim 26$  and 33%, except for the high-  
109 strain OR56 sample of  $\sim 74\%$ ) at a temperature of  $900^{\circ}\text{C}$  and constant strain rate of  
110 approximately  $10^{-6} \text{ s}^{-1}$  – corresponding to a constant displacement rate of  $1.5 \times 10^{-5}$   
111  $\text{mm}\cdot\text{s}^{-1}$  (Table 2). The strain rate of experiments in the rig at Orléans was slightly  
112 lower (by  $\sim 30\%$ ) than in the Tromsø rig. However, for a power law solid, this  
113 difference in strain rate should lead to only a subtle difference in flow stress.  
114 Confining pressures were 600, 700, 800, 1000, 1250, 1500 and 2000 MPa (Table 2).  
115 In the new type of solid medium apparatus, the pressure is regulated by a hydraulic  
116 pump and remains constant throughout the experiment, whereas in the conventional

117 Griggs apparatus, it is not regulated and typically increases towards the end of the  
118 experiment (the increase is corrected in the mechanical record).

119 One sample (OR56) was deformed at 2000 MPa at a faster strain rate of  $\sim 10^{-5} \text{ s}^{-1}$   
120 ( $1.5 \times 10^{-4} \text{ mm.s}^{-1}$ ) and shortened by 70%. As the hit point was reached during the  
121 pumping phase, the strength of this sample could not be determined.

122 Two strain rate stepping experiments have been performed in order to determine  
123 the n-exponent of the flow law. During these experiments, the pressure (2000 MPa  
124 for OR68 and 800 MPa for OR79) and temperature (900°C) conditions were kept  
125 constant but the strain rate was changed. For both experiments, the strain rate  
126 sequence was  $\sim 1 \times 10^{-5}$ ,  $10^{-7}$ ,  $10^{-6}$  and  $10^{-5} \text{ s}^{-1}$ . The n-exponent was then defined for  
127 each experiment as the slope between the log of the strain rates at the end of each  
128 strain rate step and the log of the differential stresses calculated by linear regression.

129 Processing of the mechanical data was carried out with a MATLAB program (Pec  
130 et al., 2016) adapted by J. Précigout following the routines of Heilbronner  
131 (<https://micro.earth.unibas.ch>). Finite strain of the deformed samples was calculated  
132 from mechanical record and from direct measurements of samples lengths before  
133 and after the experiments.

134 After the experiments, all samples were cut through the centre with a diamond  
135 saw (passing through the thermocouple positions) to obtain a longitudinal section  
136 parallel to the sigma 1 direction. Thin (30  $\mu\text{m}$ ) and thick (90 and 150  $\mu\text{m}$ ) sections  
137 have been prepared from the two parts.

## 138 *2.2 Cathodoluminescence*

139 Cathodoluminescence (CL) imaging has been carried out on samples using the  
140 light microscope (LM) and the scanning electron microscope (SEM). LM-CL-images



141 of thin sections have been recorded at ISTO (Orléans) with an OPEA cold cathode  
142 stage at approximately 10-12 kV and 120-150  $\mu$ A under low-pressure Argon gas at  
143 6.7 to 7.3 Pa. The image acquisition time was 3 to 4 s. SEM-CL-images have been  
144 recorded at BRGM (Orléans) with EDAX-Pegasus system equipped with a TESCAN  
145 CL-detector. Conditions of imaging were 5 to 15 kV and 17 to 20 mm working  
146 distance. Prior to analysis, samples were coated with 20 nm of carbon. Additional  
147 mosaic CL-images have been recorded at the University of Tromsø using a Gatan  
148 Mono-CL-system on a Zeiss Merlin Compact SEM at 15 kV and 12.5 to 13.7 mm  
149 working distance using a blue filter (380-515 nm). These images have been manually  
150 segmented and processed using ImageJ software.

### 151 2.3 Bulk sample strain analysis from sand grain shapes

152 Cathodoluminescence and plane polarized light microscopy allow to identify the  
153 original sand grains of the Tana quartzite and to distinguish them from the cement.  
154 Light microscope (cross-polarized and cathodoluminescence) mosaic images of the  
155 starting material and of samples deformed at different pressures have been manually  
156 segmented to isolate original sand grains. These sand grains have been analysed  
157 using the open-source *ImageJ* program to obtain statistics on grain parameters  
158 (cross section area, shape, orientation) as described by Heilbronner and Barrett  
159 (2013).

160 The sand grain size is expressed as the equivalent diameter ( $d_{eq}$ ) calculated from  
161 the area of grains ( $A$ ) as  $d_{eq} = 2 \times \sqrt{\frac{A}{\pi}}$ . The fabric is expressed as the aspect ratio  
162 ( $AR$ ), the mean ratio between the long ( $a$ ) and short ( $b$ ) axes of the particles ( $AR = \frac{a}{b}$ )  
163 and their orientation. The dataset has been analysed by the SURFOR method  
164 described by Panozzo (1984) in *ImageJ* for a fabric and strain analysis.

165 *2.4 Electron Backscatter Diffraction (EBSD) analysis*

166 Crystal orientations have been determined by electron backscatter diffraction  
167 (EBSD) using a Nordlys detector and an Oxford Aztec system on a Zeiss Merlin  
168 Compact FE-SEM at the University of Tromsø using a 70° tilt angle, 20 kV  
169 acceleration voltage, and 9 nA probe current. An indexing quote of 75-85% has been  
170 achieved on 4 nm carbon coated thin sections. Post-processing was performed using  
171 the MTEX v.5.2.8 program package by Bachmann et al. (2010). Orientation maps  
172 were recalculated using the 6/mmm system (hexagonal) in order to avoid displaying  
173 Dauphiné twin boundaries.

174 *2.5 Fourier Transform Infrared (FTIR) Spectroscopy*

175 Double-polished thick sections of 150 and 90 µm have been prepared from the  
176 starting material and Fourier transform infrared spectroscopy (FTIR) analysis has  
177 been performed at ISTO (Orléans) with Nicolet 6700 (Continuum, Thermoscientific)  
178 spectrometer and OMNIC (version 8) acquisition software.

179 The spectra were acquired on grain interiors, grain boundaries, and cement with  
180 64 and 128 scans and a resolution of 4 cm<sup>-1</sup> for a range of wave numbers from 5500  
181 to 1500 cm<sup>-1</sup>. A window with 40x40 µm for interior of grains and with 20x50 µm was  
182 used for grain boundary measurements (window shape and orientation was adapted  
183 to the boundaries). The background was recorded for the CaF<sub>2</sub> window carrying the  
184 sample and subtracted from the measured spectra. Water contents were calculated  
185 using the Beer-Lambert law and the Paterson (1982) calibration.

186 *2.6 Water fugacity calculations*

187 Assuming that during the experiments the H<sub>2</sub>O pressure approximatively equals  
188 the confining pressure, the water fugacity ( $f_{H_2O}$ ) can be calculated using only two pa-  
189 rameters: the temperature and the pressure. The water fugacity has been computed

190 for all experimental conditions from Tony Withers' fugacity calculator (available at:  
191 <https://www.esci.umn.edu/people/researchers/withe012/fugacity.htm>) based on the  
192 Pitzer and Sterner (1994) equation of state for water.

### 193 **3. Results**

#### 194 *3.1 Starting material*

195 For the starting material, a very pure natural quartzite from the ELKEM quarry in  
196 Austertana, Northern Norway (N70°28'39.6'', E28°32'30.1'') has been used. The  
197 rock samples come from the top part (~40 m) of the Gamasfjell formation. This  
198 sequence of late Precambrian age is approximately 300 m thick and its top part is  
199 composed of grey to white quartzite (Pevik, 2015). The uppermost part of the  
200 formation is described as very pure (> 99% of silica), and used by the aluminium,  
201 silicon, and ferrosilicon industry (Aasly et al., 2007). Chemical analysis (from ICP-  
202 EOS) published in Pevik (2015) indicate major oxide values (apart from SiO<sub>2</sub>) of  
203 3547 ± 1083 ppm Al<sub>2</sub>O<sub>3</sub>, 441 ± 365 ppm Fe<sub>2</sub>O<sub>3</sub>, and 207 ± 37 ppm TiO<sub>2</sub>.

204 The Tana quartzite underwent high grade diagenesis to very low grade  
205 metamorphic conditions and is weakly deformed in a large open antiform (Pevik,  
206 2015). However, no internal deformation on the micro-scale, no schistosity, and no  
207 lineation have been identified in the sampled rock. The blocks of the starting material  
208 collected were not oriented. The quartzite is composed of quartz grains and  
209 crystalline authigenic SiO<sub>2</sub> cement (99%, Fig. 1a) and very few accessory minerals  
210 (< 1%) such as sheet silicates (sericite, pyrophyllite, kaolinite), iron oxides (hematite),  
211 zircons, apatite, xenotime, monazite, rutile, and very rare feldspars. Due to filling of  
212 the pore space by authigenic SiO<sub>2</sub> cement, no visible porosity was detected in the  
213 light microscope or SEM.

214 Microstructural analysis and CL-imaging show that the quartz grains forming the

215 Tana quartzite are composed of (1) cores of equant and rounded detrital quartz sand  
216 grains, with various CL-colours and -intensity, and (2) a non- or dark-luminescent  
217 cement between the grains (Fig. 1c,d). The cement is in crystallographic continuity  
218 with the sand grains on which it grows epitaxially (Fig. 1a). Grains appear  
219 undeformed as they do not show any undulatory extinction, except for rare sand  
220 grains showing inherited recrystallized microstructures. A few pressure-solution  
221 contacts due to burial of the sediment are sometimes observed.

222 The size distribution of quartz grains composing the quartzite was obtained by  
223 image analysis using both polarized light and light microscopy (LM)-CL-images. The  
224 mean grain size is 204  $\mu\text{m}$  for quartz grains and 186  $\mu\text{m}$  for original sand grains  
225 (Table 1). The quartz grain sizes are larger than those of sand grains because they  
226 include the surrounding cement. The calculated mean aspect ratio (AR) is 1.56 for  
227 quartz grains and 1.55 for sand grains, indicating that grains are somewhat  
228 elongated. However, no preferred orientation of the long axes of the grains can be  
229 detected. For this reason, samples have been cored without reference to any specific  
230 orientation.

231 Electron Backscatter Diffraction (EBSD) measurements of grains from starting  
232 material Tana quartzite (one point per grain,  $n=500$ ) confirm a random fabric (Fig. 1b,  
233 with  $J=1.33$  and  $M=0.027$ ), close to a random distribution (Bunge, 1982; Skemer et  
234 al., 2005). The dislocation density of the material was characterized by transmission  
235 electron microscopy (TEM) as  $6 \times 10^{12} \text{ m}^{-2}$ , which is a typical, low dislocation density  
236 of relatively undeformed natural silicates (McLaren, 2005).

237 The mean water content calculated is  $1659 \text{ H}/10^6 \text{ Si}$  for grain interiors and  $2120$   
238  $\text{H}/10^6 \text{ Si}$  for grain boundaries.

239 *3.2 Mechanical data*

240 *3.2.1 Stress strain curves*

241 The samples show a systematic decrease in strength with increasing confining  
242 pressure (Fig. 2): the maximum final differential stress is nearly 600 MPa for a  
243 sample deformed at 600 MPa confining pressure, whereas for the sample deformed  
244 at 2000 MPa it is  $\sim 100$  MPa. This behaviour is reproducible for the two different  
245 apparatus (considering a  $\pm 30$  MPa accuracy of the solid medium apparatus;  
246 Holyoke and Kronenberg, 2010) used for a confining pressure of 1000 MPa (samples  
247 546LN and OR32; the strain rate is slightly lower in OR32) whereas for lower  
248 pressure at 600 MPa, the sample deformed in Orléans (OR60) is slightly stronger  
249 than the one in Tromsø (542LN), despite the slightly slower strain rate.

250 *3.2.2 Strain rate stepping experiments*

251 Two strain rate stepping experiments were performed at high (OR68, 2000 MPa)  
252 and low (OR79, 800 MPa) pressures to determine the n-value, or the stress  
253 exponent, of the flow law. Both experiments were conducted at 900°C with 0.1 wt.%  
254 water added and with a strain rate sequence of  $\sim 10^{-5}$ ,  $\sim 10^{-6}$ ,  $\sim 10^{-7}$  then again  $\sim 10^{-5}$   
255  $\text{s}^{-1}$ . The stress-strain-curves are presented in Fig. 3a and flow stresses for each step  
256 are reported in Table 3.

257 The initial step at  $\sim 10^{-5} \text{ s}^{-1}$  has been repeated at the end of the experiments for a  
258 test of reproducibility. The flow stresses associated to that rate are systematically  
259 higher during the final step and steady state has not really been reached for the  
260 lower confining pressure experiment. For the steps at  $\sim 10^{-7} \text{ s}^{-1}$ , both samples show  
261 an unstable and variable behaviour, probably due to the low sample strength, so that  
262 possibly some friction effects at the pistons start to play a role. In addition, at such  
263 low strain rates the initial “friction” of the run-in curve is partially recovered, so that  
264 the recorded stresses may be too low (see explanation of this effect in Tarantola et

265 al., 2012, appendix). In the next generation Sanchez apparatus, very low oil flow is  
266 injected by the hydraulic pump (0.0007 mL/min) in order to obtain this strain rate, this  
267 variation may partially reflect changes in the ambient conditions (daily variations  
268 recorded for the force, influenced by room temperature or cooling temperature  
269 variations).

270 The n-exponent is defined as the slope of the linear regression in the log-log plot  
271 of the flow stresses vs. strain rate (Fig. 3b). For the low pressure experiment (OR79,  
272 800 MPa),  $n \sim 1.42$  is obtained, and for the high pressure one (OR68, 2000 MPa),  
273  $n \sim 1.40$ . Given the uncertainties of the stress determination in the solid medium  
274 apparatus, these n-values are identical. If the slowest strain rate steps are omitted as  
275 somewhat less reliable (see above), the resulting n-values are 2.33 for 2000 MPa  
276 and 1.96 for 800 MPa confining pressure.

### 277 *3.3 Microstructures*

#### 278 *3.3.1 Light microscopy*

279 The local strain distribution in all samples is inhomogeneous. Typically, the centre  
280 and/or bottom parts of the samples are more strongly deformed (Fig. 4). The  
281 partitioning of deformation typically is detected by more elongated grain shapes in  
282 the higher strained regions (Fig. 4b). The greater elongation of individual quartz  
283 grains usually is accompanied by an increase in the amount of recrystallized  
284 material. Original quartz grains in deformed samples show evidence for plastic  
285 deformation, such as undulatory extinction, deformation lamellae and, in some  
286 places, development of subgrains (observed in the LM; Fig. 4c). Some of the  
287 progressive subgrain rotation leads to the formation of core-mantle structures (Fig.  
288 4b,c).

289 This strain gradient is slightly more pronounced for samples deformed in the

290 conventional Griggs apparatus at Tromsø (where more of the deformation is  
291 localized at the lower part of the samples) than for those deformed in the new  
292 apparatus in Orléans (where more of the deformation is located in the central part of  
293 the sample). This type of strain localisation is common in samples deformed in the  
294 solid medium apparatus (e.g., Heilbronner, 2002; Heilbronner and Tullis, 2002;  
295 Stünitz et al., 2017) and is related to the temperature gradient in the sample. The  
296 thermocouple position indicates the highest temperature of the sample ( $< 90^{\circ}\text{C}$   
297 temperature difference between sample ends and the hot zone = thermocouple  
298 position) and this region typically corresponds to the higher strain regions.

### 299 3.3.2 Cathodoluminescence

300 SEM-CL and LM-CL images show that in all deformed samples the original sand  
301 grains can still be identified by the same variation of luminescence and tints as in the  
302 starting material (Figs. 1 & 5). The sand grains in the more highly deformed parts of  
303 the sample show a more elongated shape with the long axis at a large angle or  
304 normal to the shortening direction. In the cemented area between the original sand  
305 grains, the darker luminescent cement is clearly observable. However, in the  
306 deformed samples, a new material with a bright luminescence appears in CL-images  
307 (Figs. 5 & 6). This bright luminescence material invariably has a blue colour, so that  
308 a blue filter was used in SEM-CL images to enhance its presence. The blue  
309 luminescence is not short-lived but permanent.

#### 310 3.3.2.1 Morphology of the bright luminescent material

311 The bright luminescence material in deformed samples is often concentrated in  
312 the boundary regions of grains, so that these bright regions appear predominantly in  
313 the cement at the grain boundaries of quartz grains. In some cases the bright  
314 luminescing material cuts across original sand grains (Fig. 6a,b). In the low strain  
315 parts of the samples, the brightly luminescing zones follow a clear crack morphology

316 (Figs. 5 & 6). The bright luminescent material occurs in the cracks that cut through  
317 both, sand grains and cement. This is observed in samples deformed at low  
318 confining pressure (e.g., OR42, 800 MPa, Fig. 6a) and at higher confining pressure  
319 (546LN, 1000 MPa, Fig. 6b). In samples deformed at low confining pressure, cracks  
320 are subparallel to the loading direction and appear to have been dilatant, mode I  
321 cracks, before new quartz material has filled the cracks. The inner parts of these  
322 cracks can be filled with non-luminescent material (Fig. 6a). From the boundaries of  
323 the healed mode I cracks, very thin cracks extend perpendicular to the mode I  
324 cracks, i.e., in an orientation typical for unloading cracks (Fig. 6a). The appearance  
325 of such cracks has first been observed in samples deformed in the Griggs-type  
326 apparatus by Fitz Gerald et al. (1991), and these features have been termed "step-  
327 ladder cracks". In all deformed samples, many thin cracks with bright luminescence  
328 in a direction perpendicular to the shortening direction extend from grain boundaries  
329 and cracked regions and have formed as unloading cracks during decompression of  
330 the samples (e.g., Fig. 6b).

331 In samples deformed at higher confining pressure, cracks tend to be thinner, more  
332 irregular, and are distributed more pervasively across the samples (Fig. 5e,f). They  
333 often (but not necessarily) follow the grain boundaries (intergranular cracks). For all  
334 the cracks in the samples, no or very little displacement has been observed along  
335 the cracks.

336 The brightly luminescing material sometimes forms overgrowths with crystal faces  
337 on pre-existing large grains or filling intergranular spaces (Fig. 6c,d). EBSD maps  
338 performed on these areas indicate that these faceted overgrowths are in  
339 crystallographic continuity with the parent grain. The faceted crystals can be  
340 surrounded by non-luminescent material in pore spaces (Fig. 6c,d).



341 Larger regions of bright luminescent material consist of aggregates of many small  
342 new grains (Fig. 6d). Many of the small grains are brightly luminescent throughout  
343 but others have a darker luminescent core with a bright rim of variable thickness (Fig.  
344 6d). Comparison between bright luminescent zones and the corresponding regions  
345 under crossed polarizers indicate that all of these regions consist of small grains,  
346 luminescing in blue, but not all luminescent regions consist of small grains. The  
347 amount of brightly luminescent material increases substantially with increasing strain  
348 (see also below). In a sample deformed to 74% strain (OR56) the original grains are  
349 strongly elongated and are embedded in a matrix of 30% luminescent material (Figs.  
350 5f & 12).

### 351 3.3.2.2 *Evolution of luminescent regions as a function of pressure and sample* 352 *strain*

353 For four samples (546LN, OR42, OR57 and OR56), SEM-CL longitudinal  
354 transects have been recorded from the top to the bottom (example for OR57 sample  
355 on Fig. 7). All transects were segmented manually in order to separate the original  
356 sand grains and cement from the bright luminescent material. The segmented  
357 images were then processed with the *ImageJ* software in order to quantify the  
358 amount of bright luminescence regions in samples.

359 Domains with higher finite strain show a higher proportion of brightly luminescent  
360 areas. This relationship can be observed at the scale of a given sample, between the  
361 top – relatively undeformed part – and the bottom part – where deformation is much  
362 more intense (OR62, Figs. 5e & 8a). The relationship can also be observed in  
363 samples deformed to different amounts of total strain (e.g, OR56; Fig. 5f). There is a  
364 relationship between the amount of brightly luminescent material and confining  
365 pressure demonstrated for the three samples strained up to 30% (OR42, 546LN and  
366 OR57). The amount of brightly luminescent material increases in the middle to

367 bottom part of the samples (more deformed regions; Fig. 8a): 9.88% for sample  
368 OR42 (800 MPa), 10.72% for sample 546LN (1000 MPa), 15.08% for sample OR57  
369 (2000 MPa; Fig. 7). Thus, there is a trend of increasing amount of brightly  
370 luminescent material with increasing confining pressure and with strain (Fig. 8b).

### 371 *3.4 Strain analysis from fabric and grain shapes*

372 The starting material and samples deformed at 700, 1000, 1500 and 2000 MPa  
373 confining pressure have been studied for their grain shapes and fabric by image  
374 analysis. The goal of the analysis was to compare the bulk sample strain determined  
375 from the mechanical record and from measuring sample lengths with an analysis  
376 based on strain of individual quartz grains. Such an analysis is difficult in normal  
377 quartz grain aggregates because the regions of dynamic recrystallization make it  
378 difficult to identify original grain shapes as passive markers. The outlines of original  
379 sand grains in the Tana quartzite represent true strain markers for the deformed  
380 samples, because dynamic recrystallization typically affects the grain boundary  
381 region first (e.g. formation of “core and mantle” structures), i.e. the cement regions.  
382 However, the sand grain outlines are located inside the original quartz grains and are  
383 only rarely affected by the bright luminescence regions. Furthermore, they present  
384 passive markers because the cement overgrowth is in crystallographic continuity with  
385 the original sand grains. In addition, the orientation of long axes of the sand grains  
386 and their fabric are more or less random in the starting material (Table 1).

387 An example of the segmentation of the OR62 sample (2000 MPa) is shown in Fig.  
388 9. Sand grains are manually separated from surrounding cement in LM-CL-images.  
389 In this way, only the original sand grains and their internal plastic deformation are  
390 considered for this strain analysis.

391 Sand grain size and parameters calculated from the analysis of mosaic LM

392 cathodoluminescence images are presented in Table 4. The number of analysed  
393 grains varies with the grain sizes (sample size for 546LN is smaller). The equivalent  
394 diameter of sand grains varies with the layers in the starting material and is not a  
395 function of deformation. The aspect ratio of individual grains increases for deformed  
396 samples (from 1.55 to 2.05) in comparison to the starting material. No correlation  
397 emerges between confining pressures and equivalent diameters or aspect ratios  
398 neither with sample bulk strain (calculated or measured).

399 The SURFOR program facilitates the calculation of the sample strain/fabric  
400 anisotropy from the particle outlines of the starting material and deformed samples.  
401 The results are presented in Table 5 and shown in Fig. 10. The difference between  
402 the minimum and maximum of the projection curves corresponds to the global strain  
403 value (from 0 to 1). The angular difference between the maximal and minimal  
404 position of the projection curve should be  $90^\circ$  (corresponds to the angle between  
405 shortening and extension direction), and the value of the minimum corresponds to  
406 the shortening direction.

407 For the starting material (TQ) the minimum of the curve is 0.965 for  $\alpha=70^\circ$ ,  
408 indicating a very slight flattening of the grains. However, as the studied thin section  
409 was not oriented in a particular way and the cores for experimental samples were not  
410 made in the same orientation, this value indicates the general fabric anisotropy of the  
411 starting material but does not correspond to the orientation of this material in the  
412 apparatus. Yet, we should consider the  $\pm 3.5\%$  anisotropy as a mean error for  
413 strained samples.

414 For deformed samples, the bulk shortening for the grain fabric calculated by the  
415 SURFOR analysis is between 22.9 and 37.2%. For the samples OR52, OR64 and  
416 OR62 the values are very close to the ones calculated from mechanical data and

417 measured on thin sections and are within the error range ( $\pm 3.5\%$ ). However, for  
418 546LN, the shortening value is underestimated and overestimated for OR32  
419 (difference up to  $> 8\%$ ).

### 420 *3.5 EBSD maps and misorientation calculations*

421 Two regions were selected for detailed EBSD analysis: (1) a low strain region with  
422 limited recrystallization and formation of discrete luminescence in response to  
423 cracking (546LN), and (2) a high strain region affected by more extensive  
424 recrystallization/luminescent material (OR56).

425 In region (1) the cracks are visible in the SEM-CL image, cutting through original  
426 quartz sand grains as well as cement (Fig. 11a). The corresponding EBSD map of  
427 this region shows small new grains (clasts) that make high angle boundaries with the  
428 larger quartz grains in the traces of the cracks. In addition, some low angle  
429 boundaries also separate the small clasts from the host quartz grains (Fig. 11b). The  
430 size of the clasts (new grains) can be as small as  $1\text{-}2\ \mu\text{m}$  and as large as  $\sim 10\ \mu\text{m}$   
431 (Fig. 13). Low angle new grains (in Fig. 13 marked as “subgrains”) and the high  
432 angle new grains (in Fig. 13 marked as “recrystallized grains”) do not show different  
433 size distributions, only high angle new grains are more frequent (Fig. 13). On the  
434 scale of the EBSD maps, the smaller and larger clasts all have more or less rounded  
435 shape (Fig. 6c,d). This is the case in light microscope images, too (Fig. 4). The clasts  
436 typically have a bright blue luminescence colour. The clasts with low angle  
437 boundaries ( $< 10^\circ$ ) only show a weakly preferred misorientation axis in  $[0001]$ ,  
438 whereas the larger angle boundaries are misoriented with axes in  $[0001]$  and  $[-12-10]$ .

439 In the high strain region (2), the original quartz grains appear bright in the SEM-CL  
440 image, and the recrystallized matrix is medium grey (blue luminescence; Fig. 12).  
441 Some relict parts of original quartz grains are present as porphyroclasts with bright

442 luminescence surrounded by a matrix of recrystallized grains (separated mostly with  
443 high angle boundaries). The corresponding EBSD map shows a large number of  
444 recrystallized grains of 1-2  $\mu\text{m}$  size with high angle boundaries in the matrix (Fig. 13).  
445 The two large relict quartz grains (labelled 1744 and 3542 in Fig. 12) show a number  
446 of subgrains with low angle boundaries ( $< 10^\circ$  misorientation from the large quartz  
447 porphyroclast) forming regions of subgrains between the porphyroclast and the  
448 recrystallized matrix. The subgrains do not show different luminescence from their  
449 host grains. The two large grains show different distributions of misorientation axes  
450 for their subgrains: [0001] (in 1744) and no preferred axis (in 3542). The high angle  
451 grain boundary misorientations (Fig. 12c,d) have rotation axes of [1-100] (in 3542)  
452 and axes between [1-100] and [-12-10] (in 1744). The subgrain sizes are in the  
453 same range as the sizes of small recrystallized grains (Fig. 13).

454 The sizes of relict grains do not differ between the two microstructures (Fig. 13  
455 and Table 6). Relict grains are those grains that are located in a matrix of  
456 recrystallized grains. The original large porphyroclasts of the starting material  
457 typically are too large to be included completely in the EBSD maps. They are cut off  
458 at the margins of the map and therefore are not counted. The proportion of subgrains  
459 and recrystallized grains is much higher in the high strain sample OR56 than in the  
460 lower strain sample 546LN (Fig. 13).

## 461 **4. Discussion**

### 462 *4.1 Deformation processes*

463 We observe two deformation processes that operate in all deformed samples:  
464 Cracking and crystal plastic deformation. Crystal plastic deformation is visually  
465 dominant in the microstructures, and the fact that the bulk sample strain can be  
466 calculated from the particle strain of original sand grains within a few percent error  
467 (Fig. 10, Tables 2 & 5) documents that most of the bulk shortening of the samples is

468 produced by crystal plastic deformation of individual original quartz grains. The  
469 undulatory extinction, deformation lamellae, and progressive subgrain rotation (Figs.  
470 4 & 12) observed in the quartz grains are consistent with this observation. Thus, the  
471 dominant process of deformation in all samples is crystal plastic deformation, i.e.  
472 dislocation creep, as it has been documented previously for quartz under these P-,  
473 T-, strain rate conditions (e.g., Griggs, 1967; Jaoul et al., 1984; Kronenberg and  
474 Tullis, 1984; Hirth and Tullis, 1992).

475 Despite the dominance of plasticity, cracking takes place in all samples (Figs. 5 &  
476 6). Even though the crack morphology changes with confining pressure, there is one  
477 feature that is common to all observed cracking: the cracks accommodate very little  
478 displacement, so that the cracks do not contribute significantly to the finite strain.  
479 Furthermore, the highest stresses attained in these experiments are all (except  
480 sample OR60) below the Goetze criterion ( $\Delta\sigma < P_{\text{conf}}$ ). This criterion defines the  
481 upper differential stress limit of plastic or viscous deformation (brittle-plastic  
482 transition; Kohlstedt et al., 1995). In the presence of a pore fluid, this criterion has to  
483 be considered with some care (Hirth and Beeler, 2015; Beeler et al., 2015). The  
484 effect of pore pressure at high pressure and temperature conditions and small  
485 amounts of pore fluid is poorly investigated, but recent results by Okazaki et al.  
486 (2021) indicate that at the porosity of our experiments (0.12 vol%, using the density  
487 of H<sub>2</sub>O from data by Larrieu and Ayers (1997)), the potentially expected reduction of  
488 differential stress by pore pressure effects should be in the range of experimental  
489 error ( $\pm 30$  MPa; Holyoke and Kronenberg, 2010). Thus, the effective pressure  
490 coefficient  $\alpha$  in  $\sigma_{\text{eff}} = \sigma_n - \alpha P$  (see Beeler et al., 2015) should be close to zero, so that  
491 dominant viscous deformation can be concluded from the mechanical data for all of  
492 the samples, consistent with the dominant deformation mechanism in the  
493 microstructures.

494 The observed mode I cracks (Fig. 6a) have geometries typical of those formed at  
495 low confining pressures ( $P_{\text{conf}}$ ), (e.g. Paterson and Wong, 2005) and their occurrence  
496 in samples deformed at the lowest  $P_{\text{conf}}$  is not surprising. We expect that such cracks  
497 in the higher pressure samples have formed during early stages of the experiments,  
498 by processes described previously in quartz experiments in the solid medium  
499 apparatus (Fitz Gerald et al., 1991; Chernak et al., 2009; Tarantola et al., 2010;  
500 Stünitz et al., 2017). These stepladder cracks (Fig. 6a,b) form as the result of a  
501 sequence of initial cracking and subsequent crack healing, during which dislocations  
502 are produced, followed by the glide of some of these dislocations. As a  
503 consequence, there is a zone of plastically shortened material that forms  
504 immediately adjacent to the crack. During unloading of the sample at the end of the  
505 experiment this zone expands and develops unloading cracks (Fitz Gerald et al.,  
506 1991; Stünitz et al., 2003, 2017). In this way the stepladder cracks document the  
507 initiation of crystal plastic deformation in which local brittle processes and healing of  
508 cracks play an important role. The CL observations document that cracking is a  
509 ubiquitous feature during the dominant plastic deformation of quartz at high  
510 temperatures of 900°C. At higher confining pressures cracks tend to have a different  
511 geometry, but the interaction of cracking, crack healing, and plastic deformation is  
512 not expected to change. The microstructures also document that cracking without  
513 major displacement and thus without major kinematic contribution to strain  
514 accommodation may be common at high temperatures when the sample strain is  
515 dominantly accommodated by plastic mechanisms and flow stresses are low,  
516 especially at high confining pressure (Fig. 2).

#### 517 *4.2 Recrystallization processes*

518 Recrystallized material in all samples is marked by blue luminescence. The blue  
519 luminescence colour is caused by a re-working (recrystallization) of quartz involving

520 an interaction with the aqueous fluid facilitating an exchange of trace elements or  
521 producing defects during crystal growth (e.g. Ramseyer et al., 1988; Götze et al.,  
522 2001). The blue luminescence is and has been attributed to Ti-incorporation into  
523 quartz (Spear and Wark, 2009; Bestmann and Pennacchioni, 2015). It is unclear how  
524 Ti-incorporation could have been achieved in the deformed samples in our  
525 experiments, because a major Ti-source is lacking (although minute amounts of Ti-  
526 phases (rutile) are present in the starting material) and Ti-incorporation during  
527 recrystallization has been demonstrated to be slow and not producing homogeneous  
528 equilibrium compositions easily (Negrini et al., 2014). A detailed analysis of the trace  
529 element exchange that causes the luminescence was not attempted here because it  
530 is beyond the scope of this study. However, as a potential origin of luminescence by  
531 deformation-induced defects is very unlikely – these defects in quartz typically show  
532 red luminescence colours (Hamers et al., 2016, 2017), and this type of luminescence  
533 disappears after some electron beam irradiation (Bestmann, pers. Comm.) – it is  
534 inferred that luminescence is caused by an interaction of the quartz with an aqueous  
535 fluid during the reconstitution of the material. Therefore, it is concluded that the  
536 luminescence observed in our samples is the result of exchange of trace elements  
537 with a fluid during recrystallization of quartz (facilitated by boundary migration). This  
538 makes luminescence a useful tool to trace recrystallized material in the deformed  
539 samples.

540 Some of the recrystallization takes place by progressive subgrain rotation as  
541 documented by EBSD maps (Figs. 11 & 12). Subgrains with misorientation angles of  
542  $< 10^\circ$  are dominantly rotated around the [c]-axis. Such rotations can be produced by  
543 tilt walls made up of edge dislocations in the prism planes with  $\langle a \rangle$  Burgers vectors  
544 or by twist boundaries in the basal plane made up of screw dislocations with  $\langle a \rangle$   
545 Burgers vectors (e.g., Trépiéd et al., 1980; Kilian and Heilbronner, 2017). Thus, this



546 constitutes evidence for prism  $\langle a \rangle$  or basal  $\langle a \rangle$  slip. For misorientations  $> 10^\circ$ ,  
547 rotation axes are parallel to  $[-1100]$  or between  $[-1100]$  and  $[-12-10]$ , consistent with  
548 basal  $\langle a \rangle$  or combined basal  $\langle a \rangle$  and prism  $\langle c \rangle$  slip. However, it is difficult to infer  
549 slip systems accurately for larger misorientation angles, because misorientations  
550 may partially be produced by grain boundary sliding processes once high angle  
551 boundaries are established. The microstructures of subgrain boundaries within  
552 original quartz porphyroclasts and the immediately adjacent grains with high angle  
553 boundaries exhibit core-mantle structures (Fig. 12) and indicate progressive subgrain  
554 rotation as the dominant recrystallization mechanism for such grains. Subgrains with  
555 low angle boundaries and recrystallized grains with high angle boundaries do not  
556 differ systematically in size (Fig. 13). The P-, T-, strain rate conditions of deformation  
557 of the samples are those of regime 2 creep according to Hirth and Tullis (1992),  
558 where rotation recrystallization is dominant. In these microstructures, the  
559 luminescence of the porphyroclasts and their subgrains does not change, so that  
560 non-recrystallized parts of porphyroclasts with subgrains still maintain the original  
561 luminescence of the porphyroclasts (Fig. 12a). This observation is consistent with the  
562 process of fluid exchange discussed by Negrini et al. (2014): formation of subgrain  
563 boundaries is a climb process, which does not *a priori* involve any grain boundary  
564 mobility and thus no interaction or exchange with a fluid phase. Once high angle  
565 boundaries are established, grain boundary mobility may lead to exchange with a  
566 fluid, producing different luminescence colours of these recrystallized grains.

567 A completely different process can form new grains in the deformed samples, too.  
568 Local cracking produces small new grains, especially at grain boundaries and in local  
569 cracks cutting through original quartz grains (Figs. 5, 6 & 11). Some new grains show  
570 high angle misorientation relationships, whereas others show low angle relationships  
571 along the same zones of cracks (Fig. 11). The CL images indicate that in some

572 cases, inner parts of clasts can still show the original luminescence of old grains,  
573 whereas rims and small grains nearby show the blue luminescence typical of  
574 reconstituted material (Figs. 6d, 11, 16). From these microstructures, it is inferred  
575 that the small clasts in cracks have mobile boundaries that migrate and can produce  
576 a reconstitution (recrystallization) of the quartz and exchange trace elements to  
577 produce the different luminescence. Local grain boundary migration processes have  
578 been inferred in previous studies (Stipp et al., 2002a, 2002b), where the migration  
579 process is inferred to be the first step. A second type of process is subsequently  
580 required to produce isolated new grains, because the local migration itself cannot  
581 isolate small new grains. This second step can be subgrain formation and/or local  
582 cracking (Stipp et al., 2002a, 2002b). In this study, we can observe the dominance of  
583 cracking in certain locations of the sample, where new grains form (Figs. 4 & 5). The  
584 sequence in this study is reversed: the cracking occurs first, followed by boundary  
585 migration to exchange with the fluid. It is proposed here that cracking and local  
586 boundary migration operate to produce microstructures that are consistent with what  
587 is termed “bulging recrystallization” in the literature (e.g., Bailey and Hirsch, 1962;  
588 Stipp et al., 2002a, 2002b; Stipp and Kunze, 2008). Local bulging may occur in these  
589 samples, too, so that probably not all of the bulging recrystallization microstructures  
590 are induced by cracking.

591 The brittle origin of new grains in quartz and other silicates during plastic  
592 deformation has been described by van Daalen et al. (1999) in natural rocks, and by  
593 Stünitz et al. (2003, 2017) and Vernooij et al. (2006) in experiments. The CL-  
594 microstructures of these deformed samples indicate that the original clasts become  
595 modified by grain boundary migration processes after their initial formation by  
596 cracking. The misorientation of the small clasts with low angle boundaries shows a  
597 weak preference of c-axis rotation, whereas higher angle boundaries can be

598 misoriented by c-, a-, or m-axis rotation (Fig. 11).

599 Grains with blue luminescence and faceted crystal shapes in open pore spaces  
600 tend to show oriented overgrowth in crystallographic continuity with original  
601 porphyroclasts (Figs. 6c & 11b) and have been described by Palazzin et al. (2018)  
602 for dilatant sites that act as small local reservoirs of fluids. These microstructures  
603 testify to the fast precipitation in dilatant sites and to the fact that there is excess H<sub>2</sub>O  
604 in these samples (so that  $a_{\text{H}_2\text{O}} = 1$ ). Thus, the samples deformed at different  
605 confining pressures show two types of recrystallization that takes place during  
606 deformation: (1) progressive subgrain rotation, and (2) crack-related local grain  
607 boundary migration that appears to be a type of bulging recrystallization. Both  
608 operate at the same time in samples, although the geometry of some cracks  
609 indicates an early origin of these.

#### 610 *4.3 Dislocation creep and stress exponent*

611 From the conclusion of crystal plasticity, the operation of climb and dislocation  
612 glide are dominant deformation mechanisms, an n-value for the stress exponent of n  
613 = 3 to 5 would be expected for climb controlled creep (e.g., Paterson, 2012). Our  
614 strain rate stepping experiments have yielded n-values of ~1.4 or ~2, depending on  
615 whether the slowest strain rate steps are considered or not. The variance of these  
616 values and the uncertainty of the slowest strain rate steps require more detailed  
617 investigation and a better data base to determine the stress exponent more  
618 accurately. However, the present determination of  $n \sim 2$  suggests values of  $n < 3$  and  
619 is consistent with observations of Fukuda et al. (2018) and Richter et al. (2018) in  
620 quartz aggregates. Both research groups have concluded a contribution of diffusion  
621 creep (including dissolution precipitation creep) to the viscous deformation of quartz.  
622 Richter et al. (2018) have observed weaker preferred orientation of quartz in fine  
623 grained aggregates, and the  $n \sim 2$  values are explained by a combination of

624 dislocation and diffusion creep deformation in their samples. The values obtained  
625 here are lower than  $n = 3$  to  $5$ , too, and therefore inconsistent with typical  $n$ -values  
626 for pure dislocation creep (Fig. 3b).

627 The starting material in the samples of this study is coarse grained, so that  
628 diffusion creep in the starting material is unlikely. However, some of the recrystallized  
629 material (in particular the cracked material in bulging-type microstructures) is fine  
630 grained, and some diffusion creep in the recrystallized material is conceivable. In  
631 addition, extensive exchange with aqueous fluid in the recrystallized material is  
632 observed in our samples, so that diffusive mass transfer processes are evident.  
633 Thus, it is inferred that the low  $n$ -values of  $n \sim 2$  in our samples may also be explained  
634 by a combination of dislocation and diffusion creep processes. A potential  
635 explanation of dislocation accommodated grain boundary sliding (disGBS), as it is  
636 inferred in olivine (e.g., Hansen et al., 2012) is unlikely, because it would require the  
637 whole material to consist of fine grain sizes.

638 As diffusion creep is grain size dependent, its contribution to the deformation  
639 process is expected to increase with an increase of fine grained material or decrease  
640 with grain growth. In most of our samples, steady state deformation is observed (Fig.  
641 2), as it is typical for the dislocation creep regime 2 as defined by Hirth and Tullis  
642 (1992). This observation suggests that an increase of recrystallized material does not  
643 increase the contribution of diffusion creep. The reason for this may be that grain  
644 growth (= migration processes) of very small grains produced by cracking takes  
645 these grains out of the diffusion creep field (producing an equilibrium recrystallized  
646 grain size). Continued microcracking may produce new very small grains, which in  
647 turn may grow. These processes appear to be in a steady state to produce steady  
648 state deformation. However, it is premature to speculate further on this aspect, and  
649 more research is required to investigate the relationship between cracking and

650 recrystallization processes in more detail.

#### 651 4.4 Effect of H<sub>2</sub>O and pressure

652 The progressive lowering of the flow stress with increasing confining pressure in  
653 our samples is consistent with the initial observations by Kronenberg and Tullis  
654 (1984) and Mainprice and Paterson (1984). When the stresses (taken at 15%  
655 shortening, where all samples reach more or less steady state flow stress) are  
656 plotted as a function of the confining pressures, a power law relationship between  
657 the flow stress and the confining pressure of the form  $\sigma_{diff} = 505522 P_c^{-1.113} \text{ MPa}$  is  
658 found (Fig. 14), similar to that of Kronenberg and Tullis (1984). However, in this  
659 study, a coarse grained quartzite instead of fine grained novaculite was used (as in  
660 Kronenberg and Tullis, 1984), confirming that the process of pressure-dependent  
661 weakening in quartz is not dependent on grain size of the starting material.

662 Kohlstedt et al. (1995) proposed that the pressure dependence of the rheology  
663 should be expressed as a fugacity term with exponent  $m$  in the quartz flow law. The  
664 log-log plot of flow stresses vs. fugacity (Fig. 15) allows a fit of the  $r = m/n$  value,  
665 which is -0.46 for our samples. The plot in Fig. 15 shows other data from the  
666 literature together with our data, but the curve fit is only shown for data of this study.  
667 The  $r = m/n$  value is similar to that of Post et al. (1996) (-0.47), Kronenberg and Tullis  
668 (1984) (-0.5, as fitted by Post et al., 1996), Chernak et al. (2009) (-0.40), and of  
669 Holyoke and Kronenberg (2013) (-0.63, for their quartzite). It should be noted that  
670 experiments by Kronenberg and Tullis (1984) and Holyoke and Kronenberg (2013)  
671 were performed at 800°C, those of Post et al. (1996) and Chernak et al. (2009) at  
672 900°C and 1.5 GPa. A linear regression to all existing data yields almost the same  
673 slope as our data (0.458 vs. 0.460), but with a much lower correlation coefficient  
674 (0.71 vs. 0.93). This situation confirms the validity and applicability of our new data  
675 set.

676 The exponent  $m$  for the fugacity depends on the  $n$ -value of the stress exponent. In  
677 this study, a stress exponent of  $n \sim 2$  is inferred, which would result in an  $m$ -value of  
678 0.92, or  $m \sim 1$ . An  $m$ -value of  $m=1$  has been determined by Fukuda et al. (2018) for  
679 their  $n$ -value of 1.7, quite consistent with our results. If  $n$ -values closer to  $n=4$  are  
680 chosen, as determined by Paterson and Luan (1990) and Gleason and Tullis (1995)  
681 and used in Hirth et al. (2001) and others, or  $n=2.7$  (or  $n \sim 3$ , Rutter and Brodie,  
682 2004), as proposed by Tokle et al. (2019), the  $m$ -value would be  $m=1.84$  or 1.24.  
683 However, the original  $m$ -value of 1 proposed by Kohlstedt et al. (1995) is consistent  
684 with our results and those of Fukuda et al. (2018) for their  $m/n$  and  $n$ -values.

685 Lu and Jiang (2018) have proposed to correct the quartz flow law for pressure by  
686 using an activation volume term. This correction would have the opposite effect as  
687 the fugacity correction, and it is only meaningful to employ such a term once the  
688 activation energy is precisely determined. The presently existing data base of  $Q$ -  
689 values shows a large scatter (see compilation in, e.g., Richter et al., 2018), so that it  
690 appears to be necessary to determine  $Q$  accurately as a first step, before an  
691 activation volume term needs to be considered. We therefore do not consider the  
692 activation volume here.

#### 693 *4.5 What is the cause of sample weakening with increasing pressure?*

694 In previous studies of quartz deformation with the presence of  $H_2O$  the weakening  
695 effect has been inferred as due to enhanced recrystallization or recovery (Tullis et  
696 al., 1979; Tullis and Yund, 1989). Subgrain boundary formation is enhanced in  
697 quartzites deformed with  $H_2O$  and grain boundary migration forms dislocation-free  
698 grains during dynamic recrystallization. Both processes are faster at higher confining  
699 pressures. Our results are consistent with this interpretation. The amount of  
700 recrystallized material increases with increasing confining pressure (Fig. 8), and this  
701 is the most obvious difference in the microstructures with increasing confining

702 pressure, apart from a different crack geometry. The recrystallization in our samples  
703 involves grain boundary migration, which, in the presence of an aqueous fluid, can  
704 be described as a process of solution and precipitation with a very short transport  
705 distance of dissolved species across the boundary region. The solubility of  $\text{SiO}_2$   
706 increases non-linearly with increasing pressure (e.g., Manning, 1994, 2018), so that  
707 enhanced boundary migration rates are likely with increasing pressure because of  
708 enhanced solubility of quartz, provided that an efficient precipitation process can be  
709 identified. The precipitation may be favoured at local dilatant sites which may form by  
710 grain boundary sliding and in which precipitation has been observed or inferred  
711 (Fusseis et al., 2009; Menegon et al., 2015; Okudaira et al., 2015; Précigout and  
712 Stünitz, 2016, Précigout et al., 2017, 2019). In fine grained material, grain boundary  
713 sliding is an important mechanism to accommodate shape changes of grains  
714 induced by plastic deformation of grains (glide of dislocations) and to adjust for the  
715 plastic strain incompatibilities as a consequence of an insufficient number of slip  
716 systems in silicates (compared to metals). The low  $n$ -values (Fig. 3) in strain rate  
717 stepping experiments are consistent with grain boundary sliding and diffusion creep  
718 components of deformation. Grain boundary sliding and local dilatancy in the form of  
719 cavitation is likely to occur in these aggregates and may cause immediate  
720 oversaturation of the fluid and precipitation of quartz. Overgrowth of quartz seams on  
721 quartz grains is observed in larger dilatant sites (Fig. 6c). Replacement of quartz  
722 material during grain boundary migration is evident from the change of CL colours in  
723 the recrystallized material, indicating boundary migration by dissolution, precipitation,  
724 and exchange of elements with a fluid.

725 It has been pointed out above that once high angle boundaries are established,  
726 the distinction of grains formed by progressive subgrain rotation and those produced  
727 from initial cracking is possible (Pongrac et al., in prep.), but it is not easy, because

728 their size is similar and they all show bright luminescence. It is inferred that in this  
729 material the weakening effect takes place as a consequence of increased confining  
730 pressure. The inference that the pressure-dependent H<sub>2</sub>O-weakening is caused by  
731 grain boundary processes is supported by the observation by Holyoke and  
732 Kronenberg (2013) that the pressure dependence of the flow stress in polycrystalline  
733 aggregates is greater than in single crystals. Consequently, these authors have  
734 attributed the weakening effect to recrystallization/recovery processes (Tullis and  
735 Yund, 1989) at grain boundaries, too.

736 The plot in Fig. 15 suggests that the pressure effect of H<sub>2</sub>O-weakening could be  
737 greater at low pressure than at high pressures. A potentially greater m-value may be  
738 caused by a change in n- or r-values in  $m = r/n$ . The determination of the stress  
739 exponent is not precise enough at this stage to decide whether a constant n-value  
740 would require an intrinsic effect of the H<sub>2</sub>O fugacity or whether the potentially  
741 changing pressure dependence of flow stress could be caused by a change in the  
742 stress exponent n.

#### 743 *4.6 Geological application*

744 The lowering of quartz flow stresses with increasing confining pressures that has  
745 been documented since the study by Kronenberg and Tullis (1984) – also  
746 documented in this study – suggests that in subduction channels at high pressure  
747 the strength of quartz dominated lithologies is expected to be very low. Low stresses  
748 for deformation in subduction zones have been inferred by Stöckhert et al. (1997,  
749 1999), Stöckhert and Renner (1998), Stöckhert (2002) and Wassmann and Stöckhert  
750 (2013), partly based on different deformation mechanisms. This study confirms that  
751 even for dislocation creep, where stresses tend to be higher than for diffusion creep,  
752 flow is expected to take place at very low stresses at high confining pressures of 2  
753 GPa and higher.



754 The recrystallization mechanism of bulging recrystallization in quartz is usually  
755 observed at low temperature conditions or the onset of crystal plastic deformation  
756 (e.g., Stipp et al., 2002a, 2002b). The interaction of cracking and local grain  
757 boundary migration indicates that some new grains in this recrystallization process  
758 can be generated by cracking. One consequence of this process is that the host  
759 control in new grain nucleation will be more difficult to determine, because small  
760 rotation of new grains is likely, (e.g., van Daalen et al., 1999), but the rotation sense  
761 and misorientation relationship is not necessarily clear and is not dependent on slip  
762 systems.

763 Cracks are abundant in high pressure samples deformed by plastic mechanisms  
764 at high temperature, in spite of the low deviatoric stress ( $\sim 100$  MPa) and the high  
765 normal stress acting on any potential fracture plane ( $P_C \sim 2$  GPa) – there is no or  
766 very little displacement on these cracks. The interaction of cracking, crack healing,  
767 and plasticity has been shown by Fitz Gerald et al. (1991) and Stünitz et al. (2017).  
768 These observations suggest that in natural rocks, microcracks may play an important  
769 role in the initiation of plastic deformation and during dynamic recrystallization, even  
770 for the deep levels of the crust where high temperatures enable viscous deformation  
771 at low deviatoric stresses.

## 772 **5. Conclusions**

773 The deformation of quartzites in the presence of 0.1 wt.% of added H<sub>2</sub>O shows  
774 decreasing flow stresses with increasing confining pressure. An r-value of 0.46 can  
775 be fitted to the fugacity data, resulting in an m-value of  $\sim 1$  for the fugacity coefficient  
776 at an n-value of  $\sim 2$ . In addition, the following observations have been made:

777 1. The dominant deformation process is crystal plasticity in original quartz grains,  
778 so that the strain analysis of quartz grains matches the bulk sample strain with an

779 error of a few percent. Dynamic recrystallization and recovery processes are  
780 observed, too, so that the dominant deformation mechanism is dislocation creep.  
781 Some cracking occurs during plastic deformation, but as the cracks do not display  
782 significant offset, their contribution to the bulk strain is negligible.

783 2. Dynamic recrystallization of the deforming quartz grains takes place by  
784 progressive subgrain rotation and by local grain boundary migration. The grain  
785 boundary migration affects small clasts produced by cracking as well as former  
786 subgrains once high angle boundaries have been established. The recrystallized  
787 material acquires a different luminescence colour (blue), which can be used to track  
788 the recrystallized material in the microstructures. The processes indicate that  
789 nucleation of new grains may take place by cracking (in addition to other processes)  
790 and appears to be an important part of a process that is termed “bulging  
791 recrystallization”.

792 3. The stress exponent  $n$  is approximately equal to 2, when some diffusive mass  
793 transfer is important during dislocation creep. This contribution is likely to be fluid-  
794 dependent or fluid enhanced.

795 4. The amount of recrystallized material increases with increasing confining  
796 pressure. It is inferred that the increasing confining pressure has an enhancing effect  
797 on the grain boundary migration rate and thus recrystallization rate.

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#### 1084 **Figures and Tables**

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1086 showing equant grains without undulatory extinction. (b) Pole figure of random  
1087 crystallographic orientation of c-axis from 500 grains measured by EBSD. (c & d)  
1088 Plane polarized light microscope-cathodoluminescence images: equant and rounded  
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1099 circles are data from samples deformed at  $10^{-6} \text{ s}^{-1}$  constant strain rate at the same  
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1136 account for the effect of Dauphiné twins. Grain boundaries are identified by black  
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1138 subgrain (defined by a misorientation between 2 and 10°) boundaries and (d) grain  
1139 (defined by a misorientation greater than 10°) boundaries. All IPF's are plotted in  
1140 hexagonal 6/mmm symmetry with the same scale for multiples of uniform (m.u.d.) for  
1141 each plot. (e) Colour key for IPF || X map.

1142 Figure 12. Cathodoluminescence (CL) image, EBSD orientation map, misorientation  
1143 map and inverse pole figures (IPF) for high strain sample OR56. (a) CL image with  
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1145 orientations coloured parallel to the X direction, sample is plotted with hexagonal  
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1147 are identified by black bounding lines and subgrains are identified by white  
1148 boundaries (see detailed insert at the top right hand corner). (c) Misorientation map  
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1150 to the mean grain orientation. Grains 1744 and 3542 are highlighted on the IPF (bold  
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1159 different confining pressures.

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1169 Table 2. Experimental conditions of deformed samples.

1170 Table 3. Flow stresses obtained and corresponding strain rates for the two strain rate  
1171 stepping experiments performed.

1172 Table 4. Results from particle analysis. The errors correspond to the standard  
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1174 Table 5. Results from SURFOR analysis. For the finite sample strain, the first column  
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1178 (2000 MPa, high-strained) samples (corresponding maps in Figs.11 & 12).

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	$d_{eq}$ ( $\mu\text{m}$ )	AR	$d_{eq}$ ( $\mu\text{m}$ )	AR
	CP-image analysis		LM-CL-image analysis	
<b>N</b>	1423		2103	
<b>Mean</b>	203.71	1.56	186.18	1.55
<b>Median</b>	198.20	1.47	178.60	1.46
<b>RMS</b>	210.78	-	196.14	-
<b>SD</b>	54.16	0.38	61.72	0.38
<b>Max</b>	451.49	3.54	476.29	4.68
<b>Min</b>	210.78	1.01	57.47	1.01

Table 2. Experimental conditions of deformed samples.

Sample name	Griggs apparatus	Temperature (°C)	Water added (wt.%)	Confining pressure (MPa)	Strain rate (s <sup>-1</sup> )	Shortening calculated (%)	Shortening measured (%)
542LN	Tromsø	900	0.1	600	1.29E-06	33.46	30.48
544LN				1500	1.37E-06	33.48	30.71
546LN				1000	1.28E-06	31.27	28.01
OR32				1000	9.32E-07	30.96	29.39
OR42				800	8.12E-07	29.71	30.83
OR48				1250	8.80E-07	28.42	27.71
OR52				700	8.26E-07	31.98	32.12
OR56	Orléans			2000	<i>not determined</i>	74.43	
OR57				2000	<i>not determined</i>	27.76	
OR60				600	7.33E-07	30.92	30.57
OR62				2000	9.29E-07	30.53	29.51
OR64				1500	9.40E-07	29.76	26.59
OR59				1000	Hot pressed (251.7h)		0.53
OR66				2000	Hot pressed (216.8h)		-
OR68		2000	Strain rate stepping				
OR79		800	Strain rate stepping				

\*error is estimated to be up to 0.8% for a measuring error of 0.1 mm

Table 3. Flow stresses obtained and corresponding strain rates for the two strain rate stepping experiments performed.

<b>Sample</b>	<b>OR68</b>	<b>OR79</b>
<b>Pressure (MPa)</b>	2000	800
<b>Temperature (°C)</b>	900	
<b>Diff. stress at <math>10^{-5} \text{ s}^{-1}</math></b>	200.6	471.7
<b>Diff. stress at <math>10^{-6} \text{ s}^{-1}</math></b>	85.4	164.3
<b>Diff. stress at <math>10^{-7} \text{ s}^{-1}</math></b>	10.0	65.3
<b>Diff. stress at <math>10^{-5} \text{ s}^{-1}</math></b>	265.2	633.4

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Table 4. Results from particle analysis. The errors correspond to the standard deviation.

<b>Sample</b>	<b>Confining pressure (MPa)</b>	<b>Number of grains analysed</b>	<b>Equivalent diameter (<math>\mu\text{m}</math>)</b>	<b>Aspect Ratio</b>
<b>TQ2</b>	Starting material	2103	$186 \pm 62$	$1.55 \pm 0.38$
<b>OR52</b>	700	2203	$174 \pm 62$	$1.95 \pm 0.64$
<b>546LN</b>	1000	1116	$230 \pm 69$	$1.76 \pm 0.54$
<b>OR32</b>	1000	1835	$204 \pm 69$	$2.05 \pm 0.73$
<b>OR64</b>	1500	2603	$172 \pm 54$	$1.81 \pm 0.55$
<b>OR62</b>	2000	3072	$181 \pm 66$	$1.85 \pm 0.57$

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Table 5. Results from SURFOR analysis. For the finite sample strain, the first column corresponds to the calculated strain and the second column to the measured strain (refer to Table 2. for details).

Sample	Confining pressure (MPa)	Minimum of the projection curve	Angle of minimum (°)	Angle(s) of maximum (°)	Fabric anisotropy (%)	Finite sample strains (%)	
						Calc.	Meas.
<b>TQ2</b>	Starting material	0.965	70	160 – 165	3.5	-	-
<b>OR52</b>	700	0.656	90	0 – 175	34.4	31.98	32.12
<b>546LN</b>	1000	0.754	85	170 – 175	24.60	31.27	28.01
<b>OR32</b>	1000	0.628	90	0 – 180	37.2	30.96	29.39
<b>OR64</b>	1500	0.722	80	170	27.8	29.76	26.59
<b>OR62</b>	2000	0.718	85	170 – 175	28.2	30.53	29.51

Table 6. Grain sizes obtained from EBSD analysis on 546LN (1000 MPa) and OR56 (2000 MPa, high-strained) samples (corresponding maps in Figs.11 & 12).

	546LN			OR56		
	Relict grains	Recryst. grains	Subgrains	Relict grains	Recryst. grains	Subgrains
<b>Arith. Mean</b>	7.967	3.789	3.854	9.030	3.323	2.683
<b>Geom. Mean</b>	7.595	3.291	2.884	8.314	2.825	2.121
<b>RMS</b>	8.613	4.282	4.879	9.853	3.824	3.871
<b>Median</b>	7.930	3.450	3.230	8.091	2.968	2.252
<b>Mode</b>	7 – 8	3.5 – 4 4.5 – 5	4 – 4.5	7 – 9	3.2 – 3.6	1.5 – 1.7
<b>Std. Dev.</b>	2.603	2.003	2.993	3.955	1.894	2.790

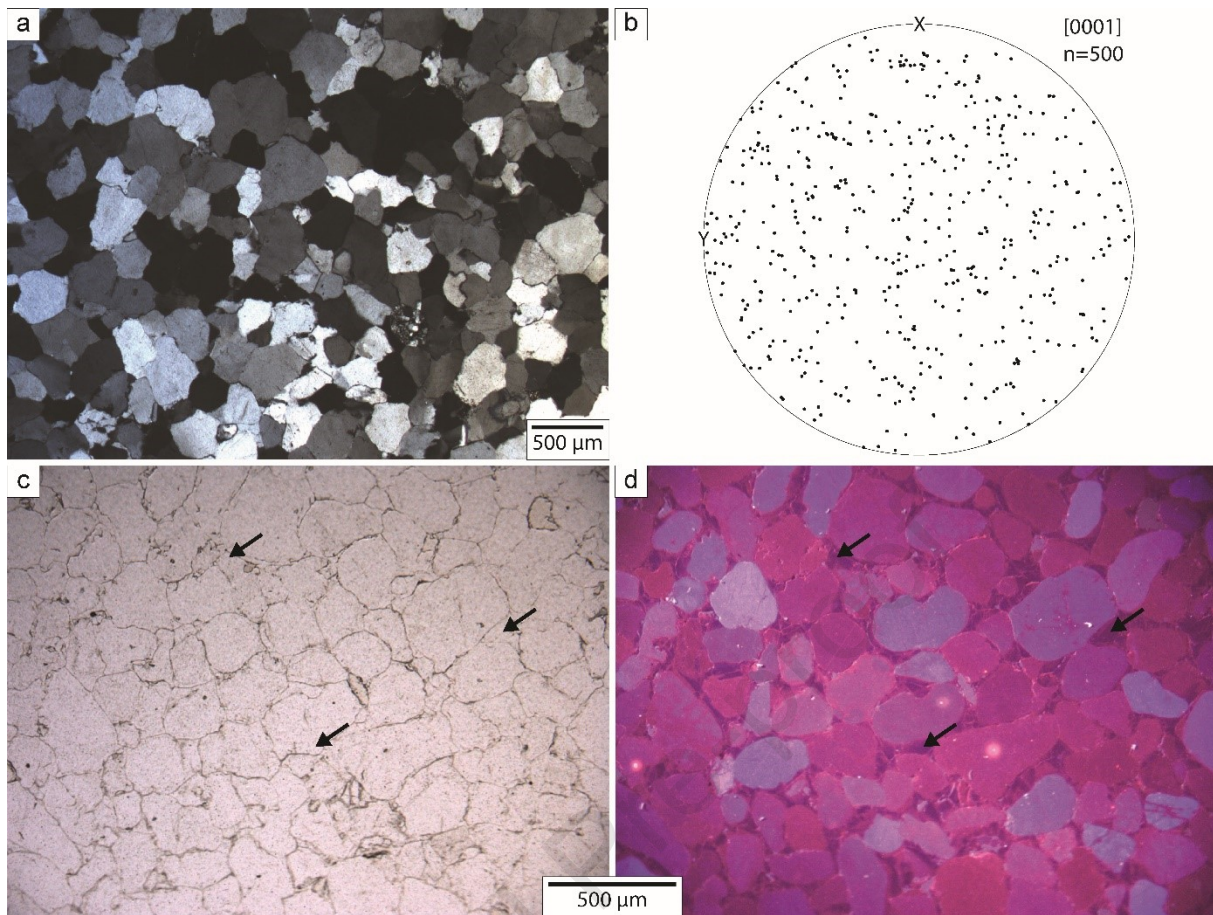


Figure 1. (a) Cross-polarized microphotograph of starting material Tana quartzite showing equant grains without undulatory extinction. (b) Pole figure of random crystallographic orientation of c-axis from 500 grains measured by EBSD. (c & d) Plane polarized light microscope-cathodoluminescence images: equant and rounded sand grains of various tints are surrounded by darker cement that grows in crystallographic continuity (arrows).



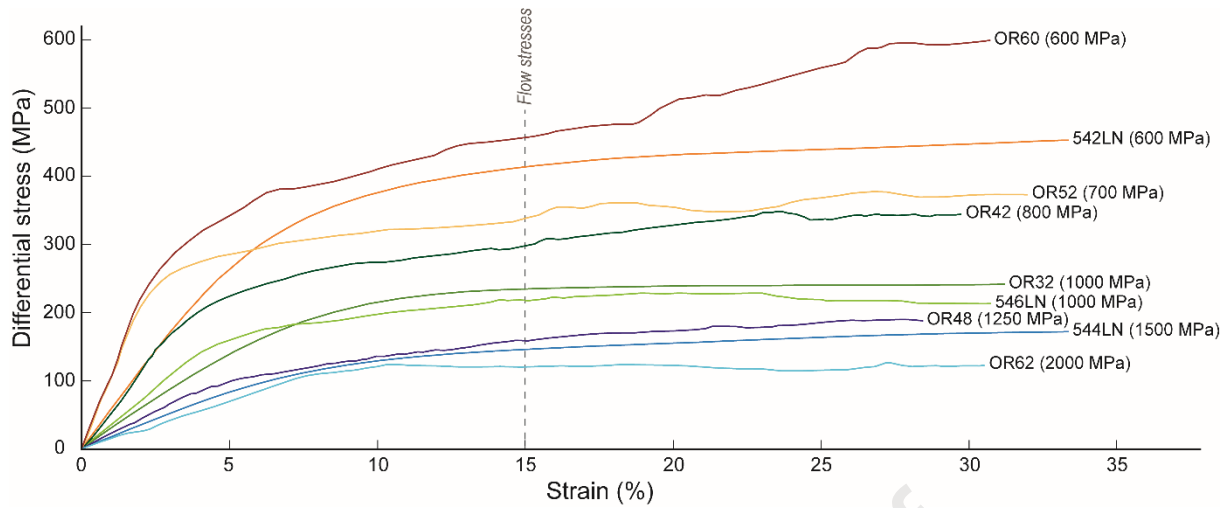


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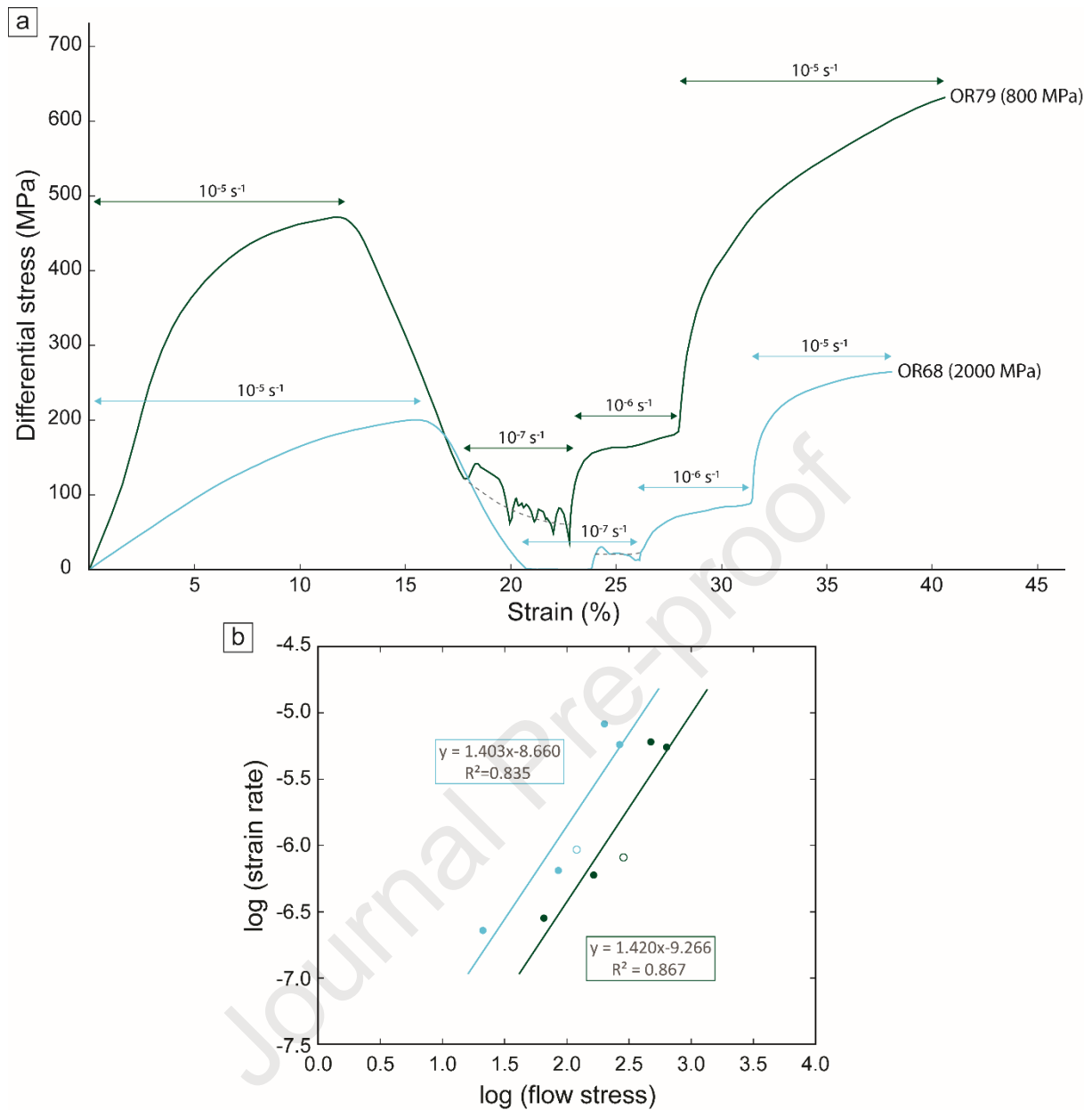


Figure 3. (a) Stress strain curves obtained for strain rate stepping experiments at 800 MPa (OR79) and 2000 MPa (OR68). (b) Plot of the log strain rate vs. log flow stress. Solid circles are data obtained from strain rate stepping experiments whereas open circles are data from samples deformed at  $10^{-6} \text{ s}^{-1}$  constant strain rate at the same confining pressures. The slopes of the regressed lines indicate  $n$ -values of  $\sim 1.89$  at 2000 MPa and  $\sim 2.06$  at 800 MPa.

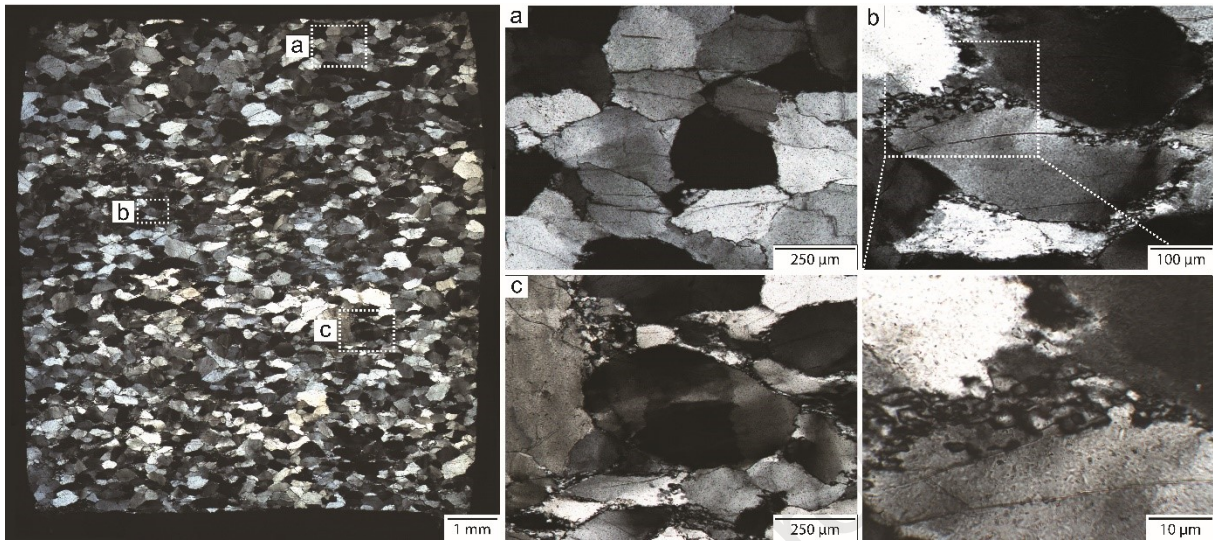


Figure 4. Cross polarized light images of sample OR32 deformed at 1000 MPa confining pressure. Left: overview of sample showing higher strain regions near the centre of the sample. In the most strongly deformed part of the sample, clasts are elongated, and limited recrystallization is observed at grain boundaries (core-mantle structures). Right: details of the recrystallized region showing undulatory extinction, subgrains, and new grains. The shortening direction is vertical.

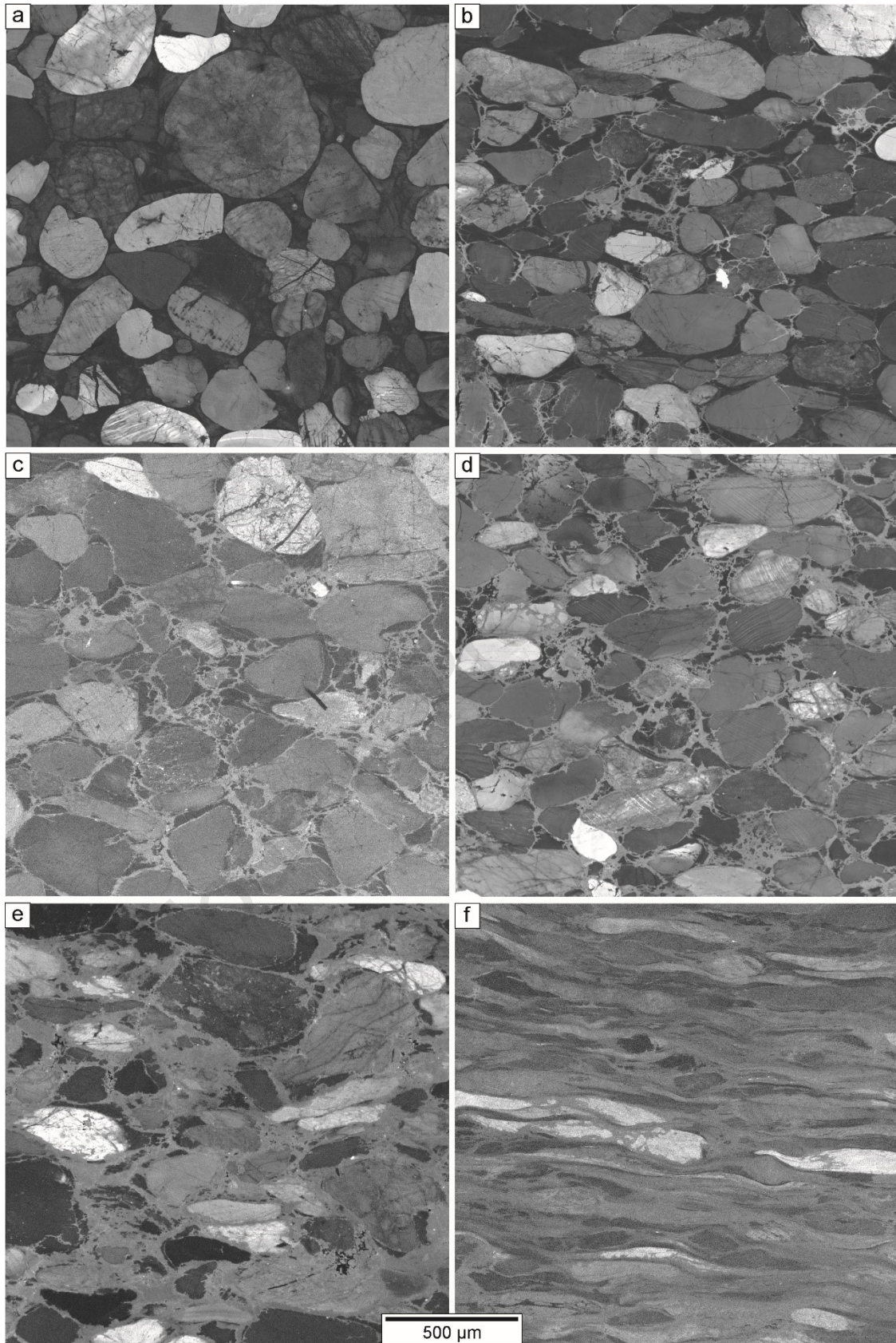


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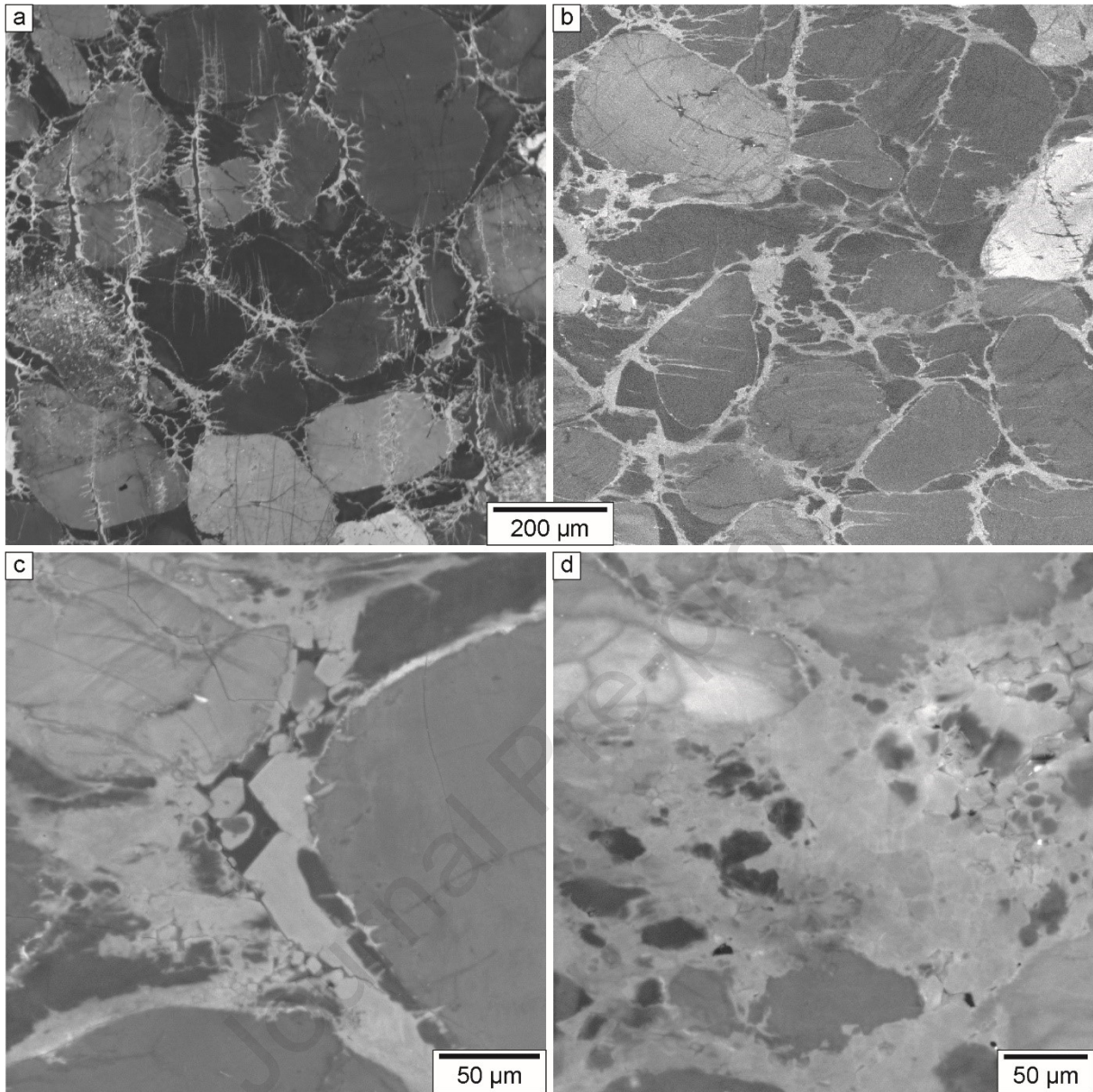


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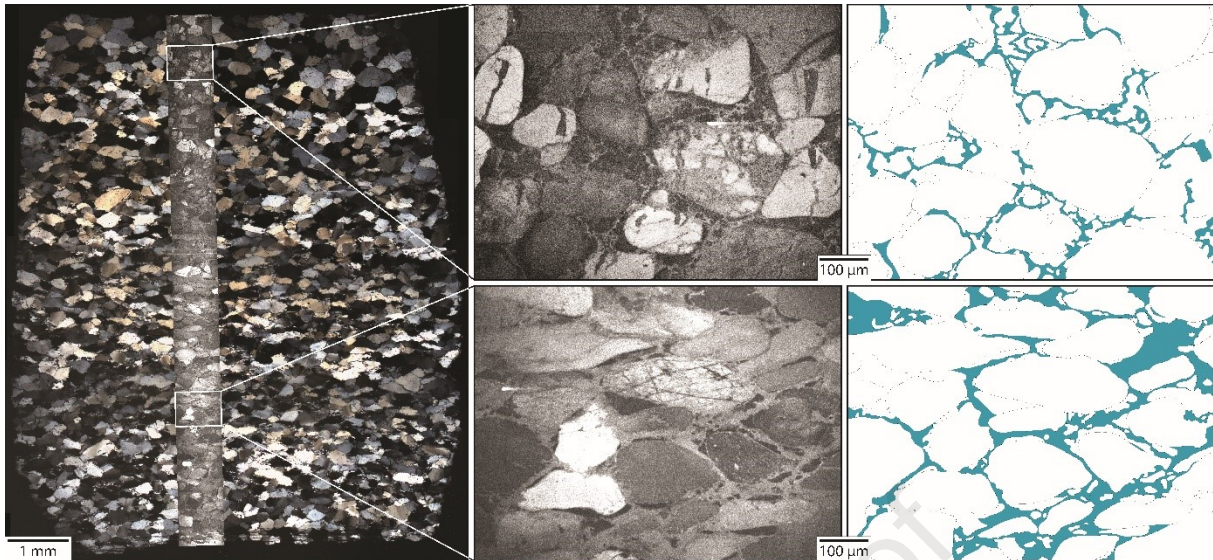


Figure 7. Left: mosaic of OR57 sample (2000 MPa) in polarized light with associated SEM-CL-images longitudinal transect. Right: examples of segmentation and identification of bright luminescent zones (coloured in blue).

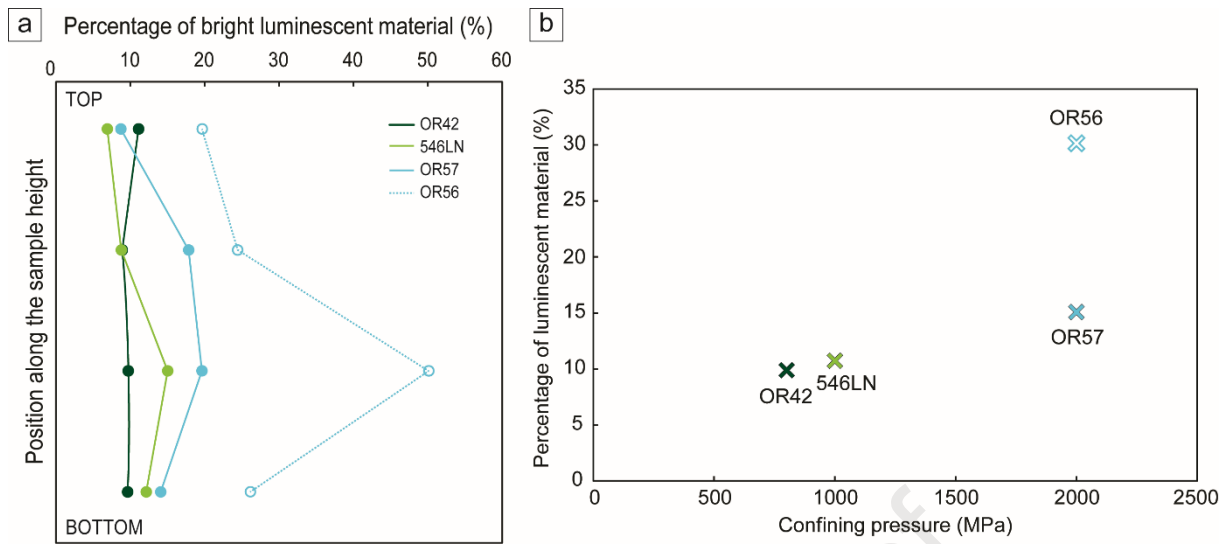


Figure 8. Percentage of bright luminescence for SEM-CL images longitudinal transects, (a) as a function of the position along the sample and (b) as a function of the confining pressure.

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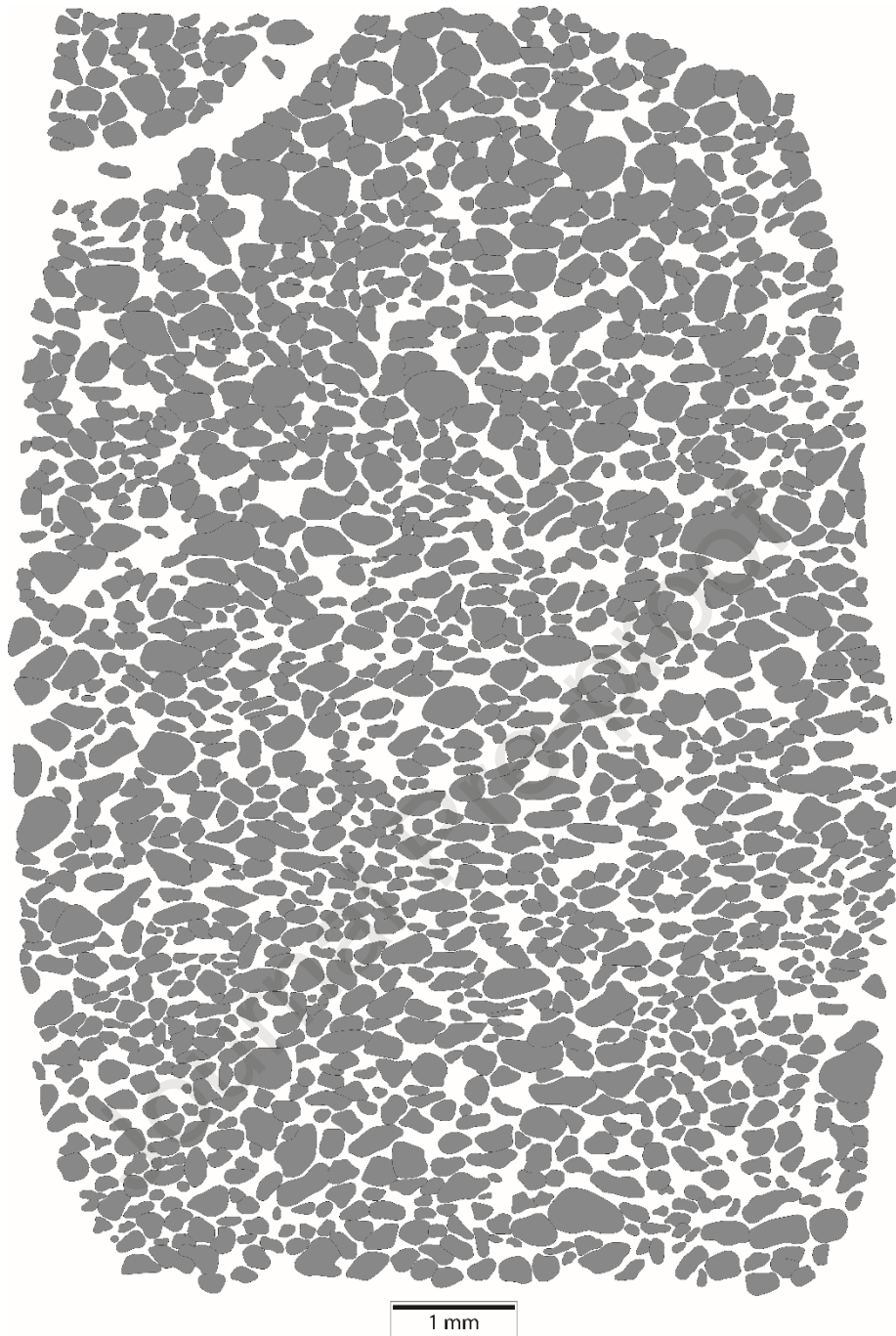


Figure 9. Example of the manual segmentation made for the OR62 (2000 MPa) thin section from optical cathodoluminescence images. Interiors of grains are isolated from the surrounding cement based on their luminescence colour contrasts. The shortening direction is vertical.



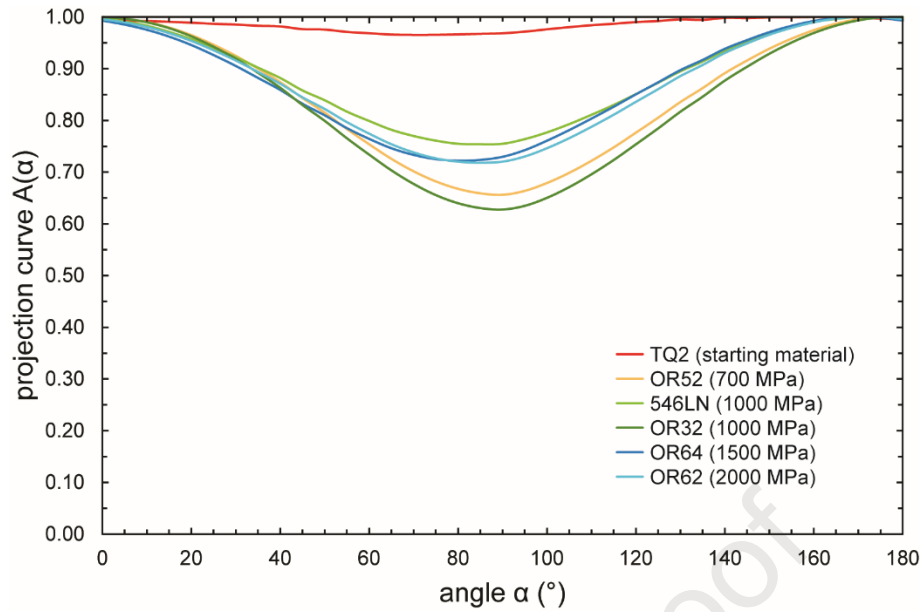


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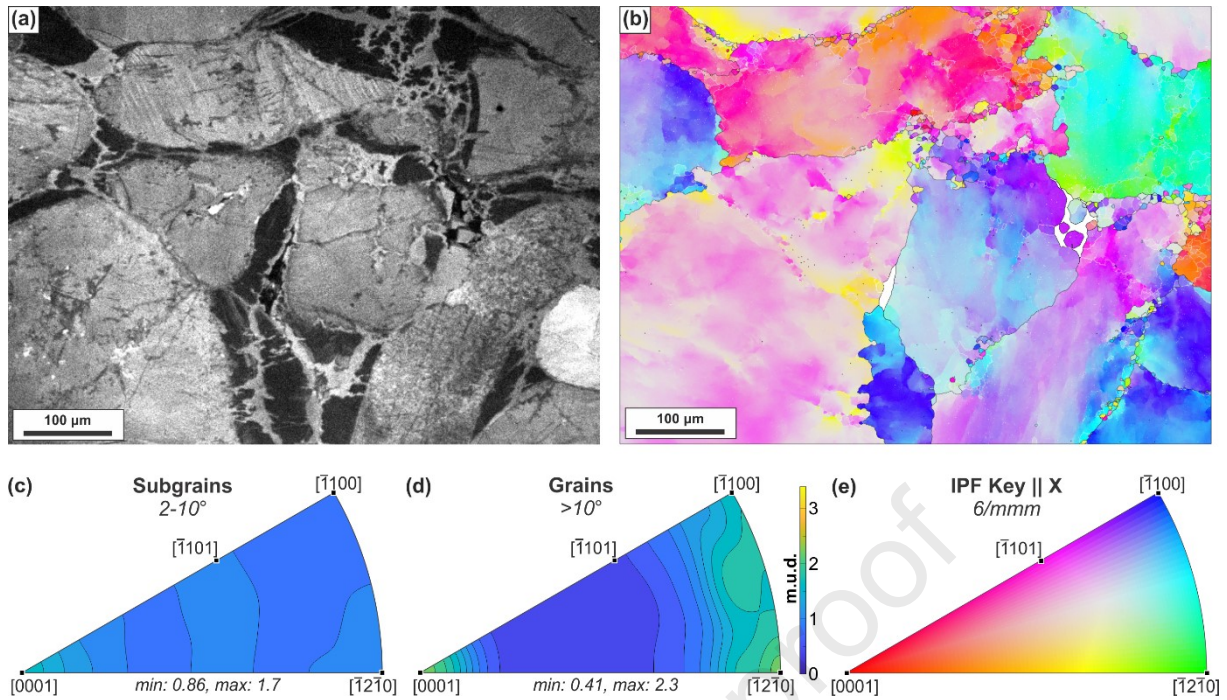


Figure 11. Cathodoluminescence (CL) image, EBSD orientation map and inverse pole figures (IPF) for sample 546LN. (a) CL image with blue (380-515 nm) optical filter to highlight recrystallized areas; (b) EBSD map of quartz orientations coloured parallel to the X direction, sample is plotted with hexagonal symmetry ( $6/mmm$ ) to account for the effect of Dauphiné twins. Grain boundaries are identified by black bounding lines and subgrains are identified by white boundaries. (c) IPF's for subgrain (defined by a misorientation between 2 and  $10^\circ$ ) boundaries and (d) grain (defined by a misorientation greater than  $10^\circ$ ) boundaries. All IPF's are plotted in hexagonal  $6/mmm$  symmetry with the same scale for multiples of uniform (m.u.d.) for each plot. (e) Colour key for IPF || X map.

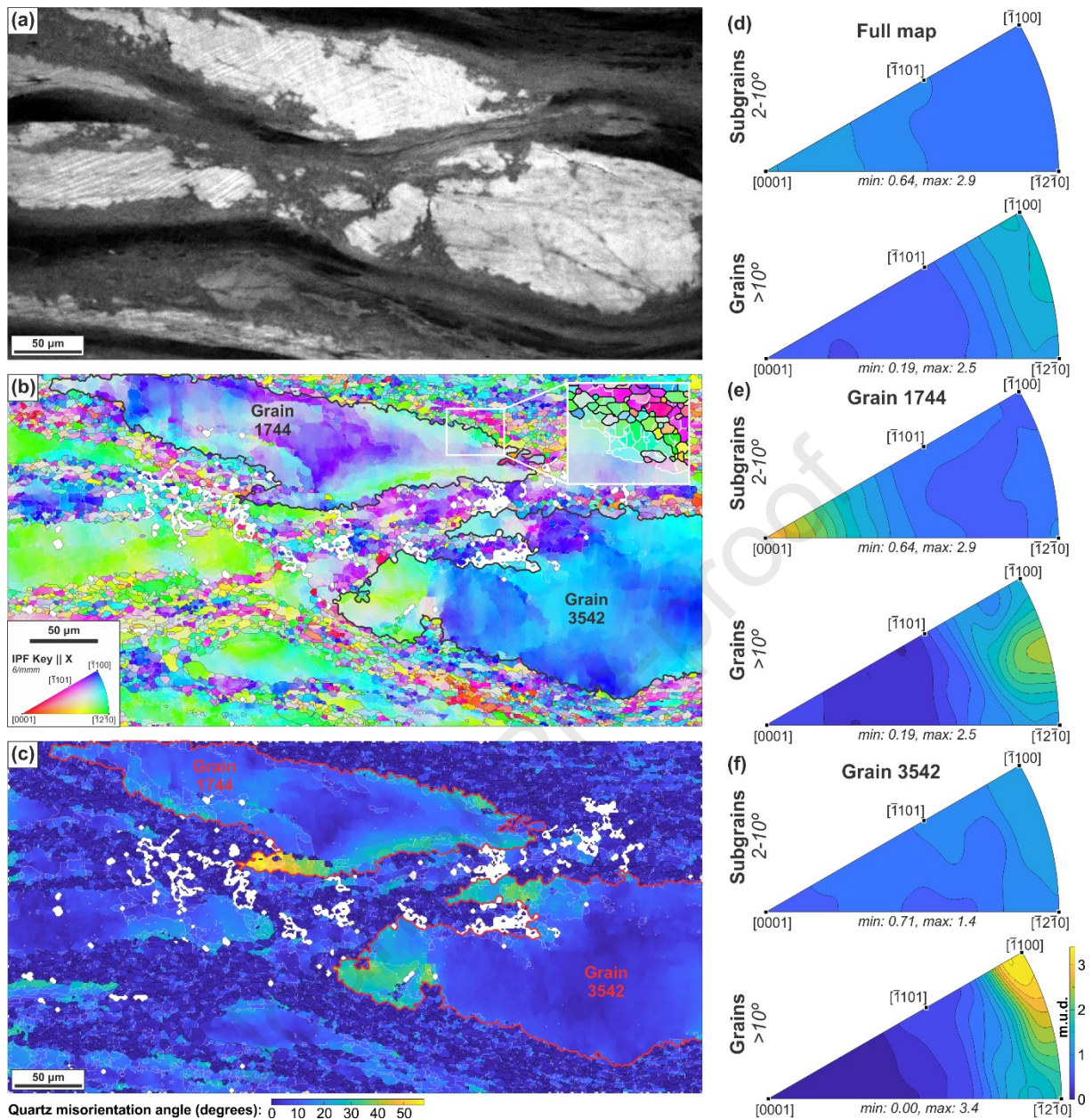


Figure 12. Cathodoluminescence (CL) image, EBSD orientation map, misorientation map and inverse pole figures (IPF) for high strain sample OR56. (a) CL image with blue (380-515 nm) filter to highlight recrystallized areas. (b) EBSD map of quartz orientations coloured parallel to the X direction, sample is plotted with hexagonal symmetry (6/mmm) to account for the effects of Dauphiné twins. Grain boundaries are identified by black bounding lines and subgrains are identified by white boundaries (see detailed insert at the top right hand corner). (c) Misorientation map where blue shows low misorientation and yellow shows high misorientation relative to the mean grain orientation. Grains 1744 and 3542 are highlighted on the IPF (bold black) and misorientation (red) maps. (d-f) IPF's for subgrain (defined by a misorientation between 2 and 10°) and grain (defined by a misorientation greater than 10°) boundaries with the d) full map, (e) grain 1744 and (f) grain 3542. All IPF's are plotted in hexagonal 6/mmm symmetry with the same scale for multiples of uniform distribution (m.u.d.) for each plot.

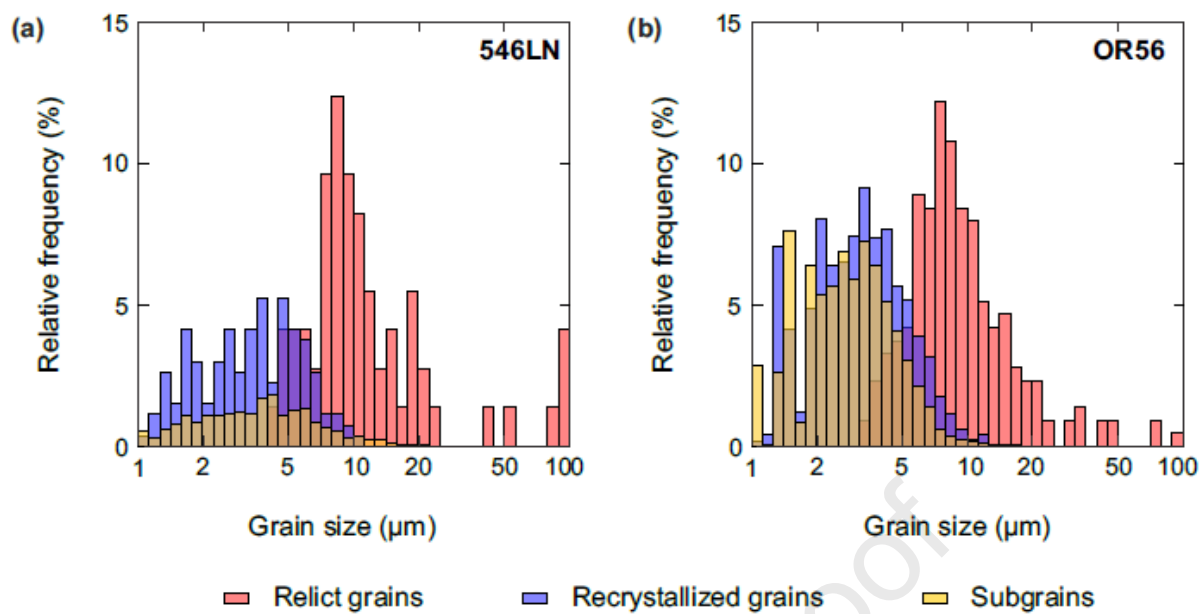


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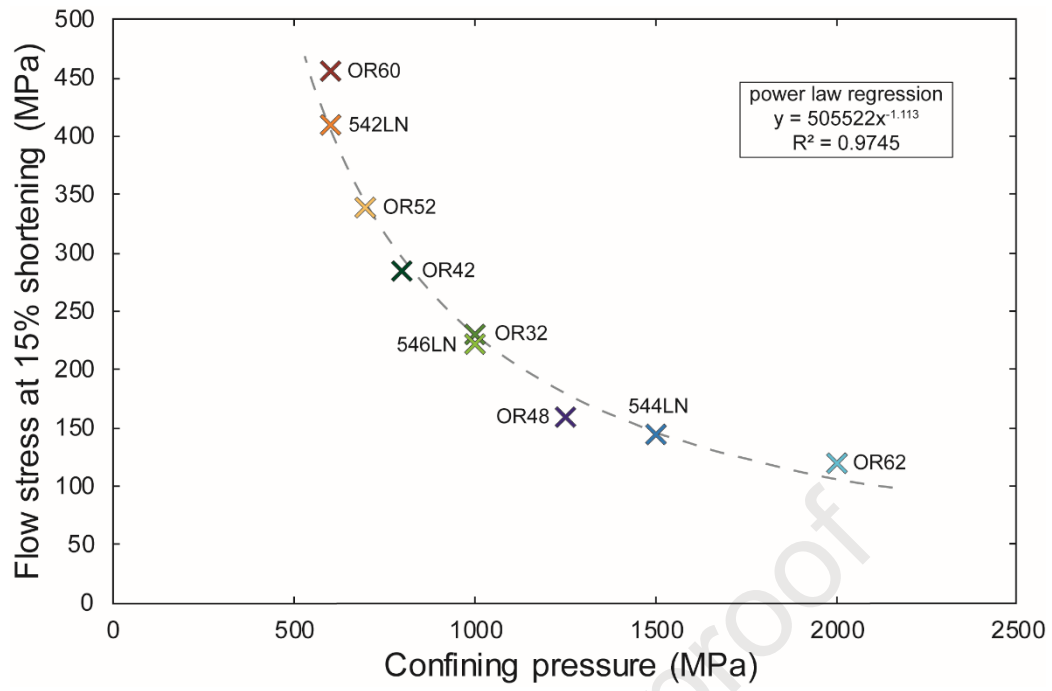


Figure 14. A power-law relationship fitted to the flow stresses (at 15% strain) at different confining pressures.

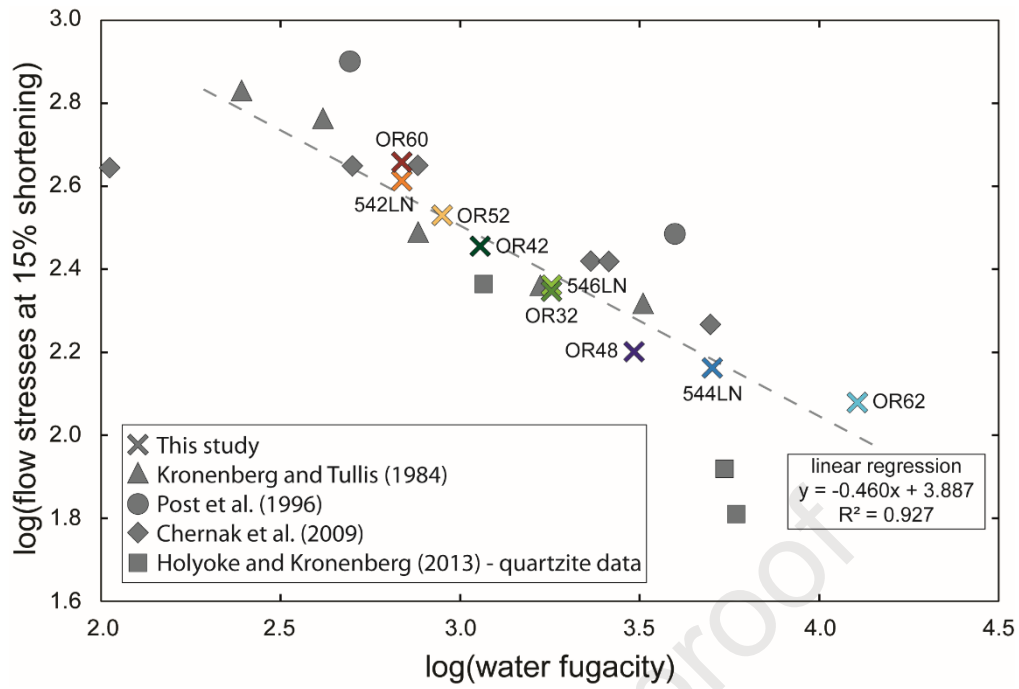
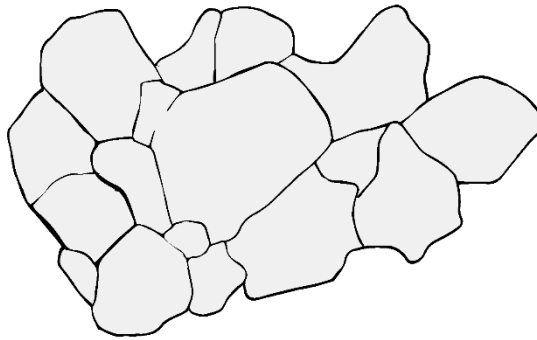
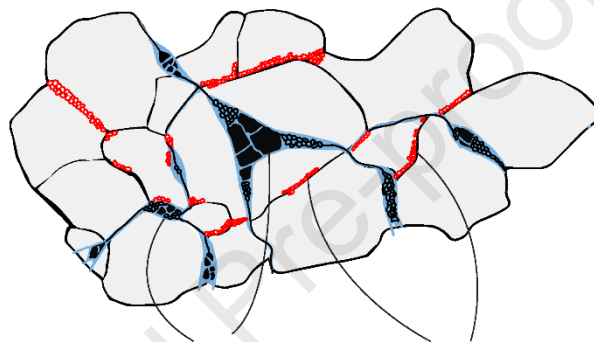


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Initial state

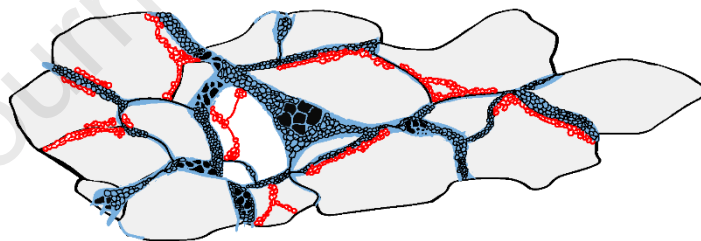


10% strain



cracking dynamic recrystallization

30% strain



 Subgrains  Cracked grains  New grains

Figure 16. Sketch of the grain assemblage evolution during deformation.

## Highlights

Inverse pressure dependence of flow stress in experimentally deformed quartzite

Bulk strain is accommodated by grain crystal plasticity

Recrystallization results from combined subgrain rotation and mode I cracking

Recrystallization processes are discriminated using cathodoluminescence of quartz

Pressure enhances grain boundary migration and reduces flow stress

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### Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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